

# International Spring School on Forefront Alloys and Advanced Materials for Extreme Conditions

15 – 17 May 2017

Sardinia, Italy

## SEVERE PLASTIC DEFORMATION TECHNIQUES FOR METALLIC MATERIALS: POTENTIALS AND LIMITS

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## Layout

- Engineering properties of metallic materials
- The **Hall-Petch** relationship
- What are and why «**severe plastic deformation (SPD)** techniques»?
- Types of SPD
- Focus on «**Equal-Channel Angular Pressing (ECAP)**» and «**High-Pressure Torsion (HPT)**»
- Potentials and Limits of ECAP and HPT (industrial scale-up)
- **Outlook**

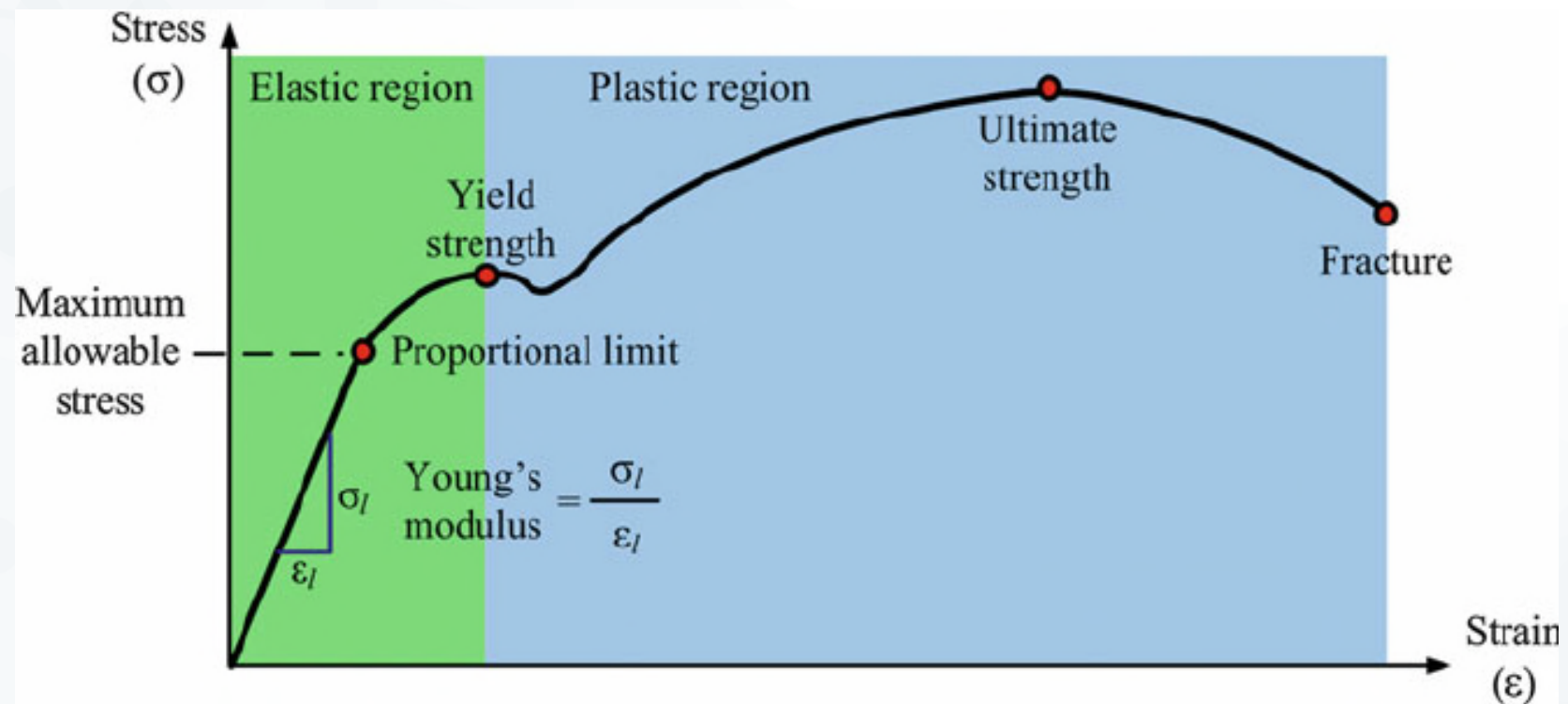
## Technological meaningful metal properties

Yield strength

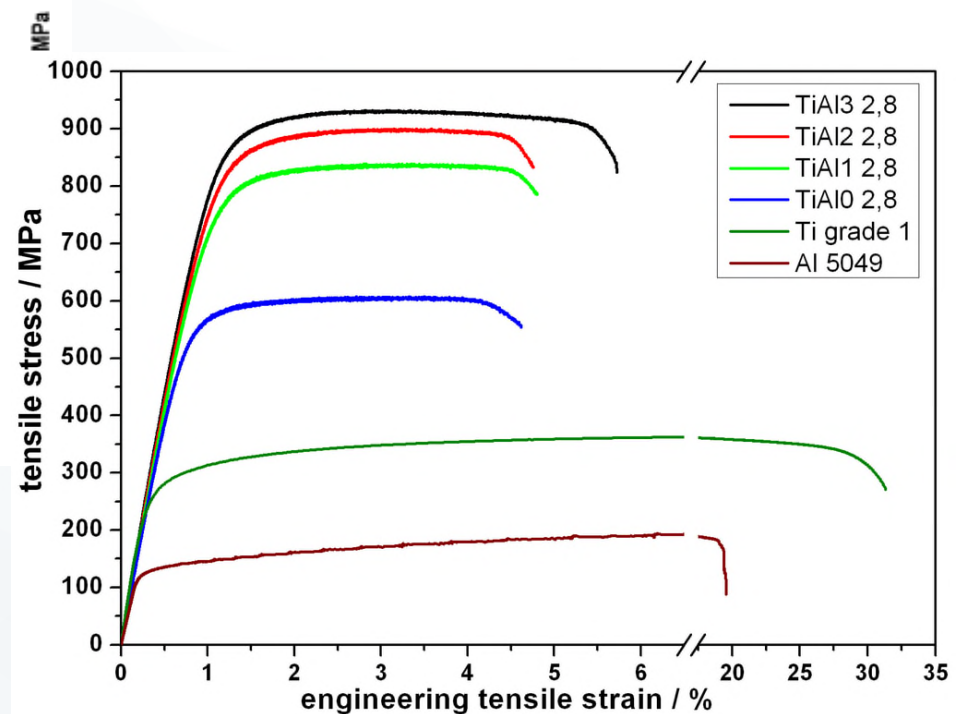
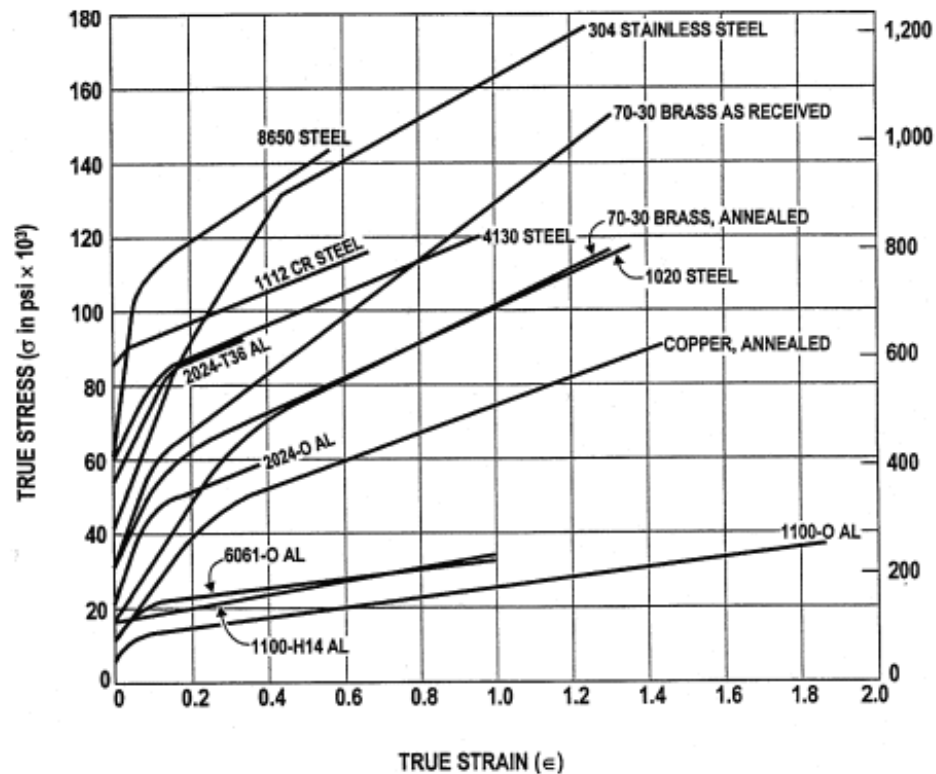
Ductility

Young's modulus (Elastic modulus)

# Technological meaningful metal properties



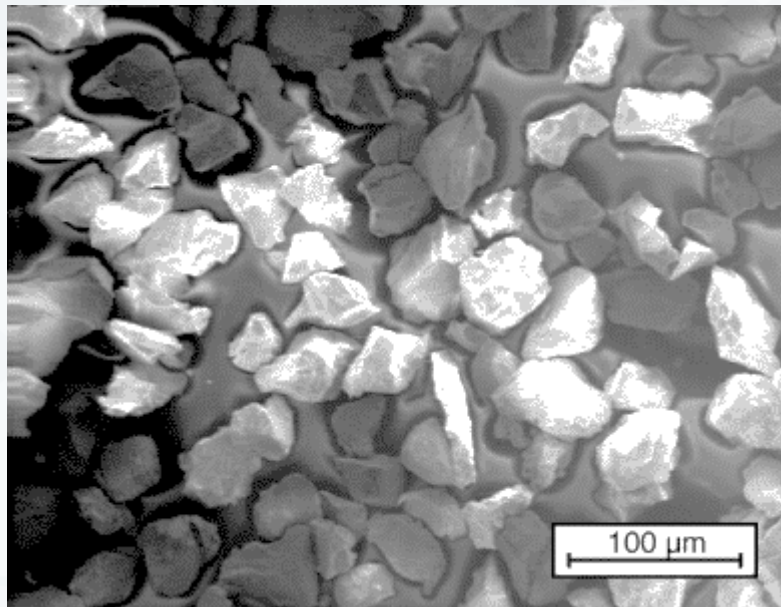
# Technological meaningful metal properties



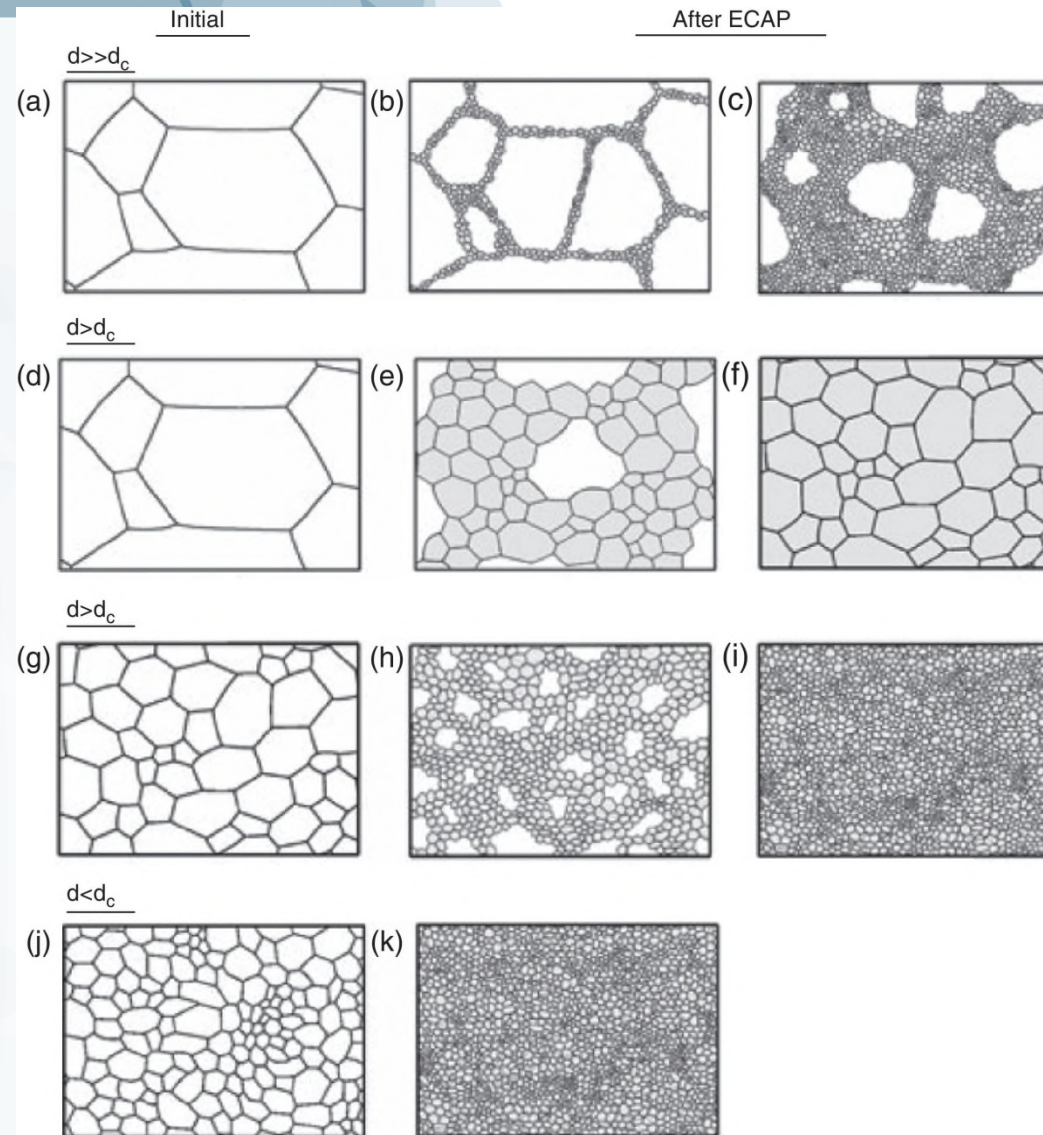
# Hall-Petch relationship

Effect of grain size on the Yield strength and Young's modulus of elasticity of metallic material

$$\sigma = \sigma_0 + K_{HP}/\sqrt{(d)}$$



# usefulness of grain size reduction

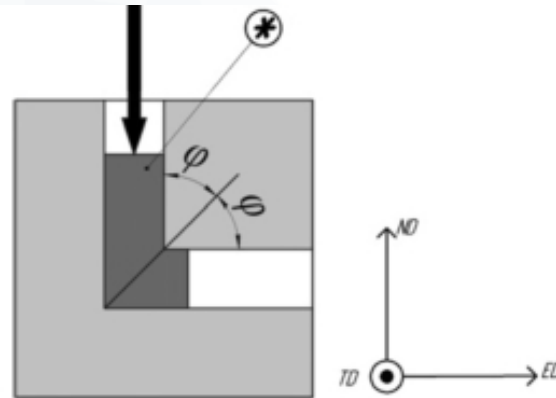


$$\sigma = \sigma_0 + K_{HP}/\sqrt{d}$$

# Severe Plastic Deformation techniques



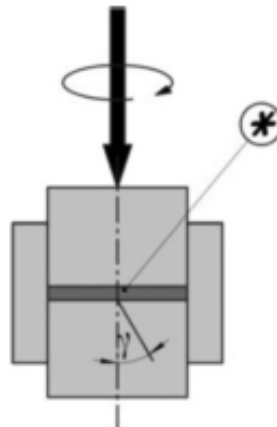
Equal-channel angular pressing (ECAP)



$$\varepsilon_{eff} = N \frac{2}{\sqrt{3}} \cot(\phi)$$

$N$ , the number of ECAP passes

High-pressure torsion (HPT)

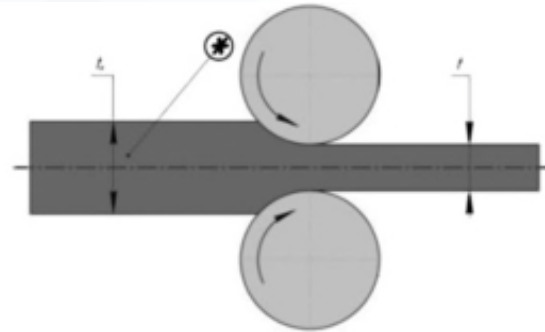


$$\varepsilon_{eff} = N \frac{2}{\sqrt{3}} \frac{\pi r}{t}$$

$r$ , the distance from the axis,  $t$ , the thickness of the sample,  $N$ , the number of revolutions

# Severe Plastic Deformation techniques

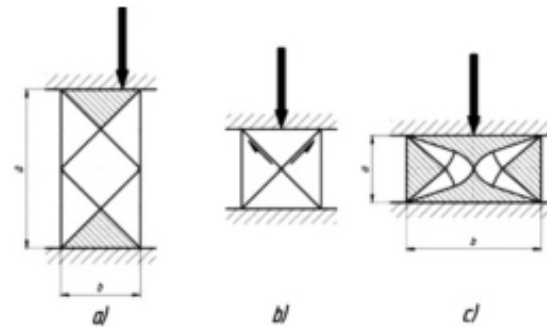
## Accumulative roll bonding (ARB)



$$\varepsilon_{eff} = N \frac{2}{\sqrt{3}} \ln\left(\frac{t_0}{t}\right)$$

$t_0$ , the initial thickness of the sample,  $t$ , the thickness of the sample after rolling,  $N$ , the number of passes

## Multi-axial forging



$$\varepsilon_{eff} = N \frac{2}{\sqrt{3}} \ln\left(\frac{a}{b}\right)$$

Strain is non-uniform.  $N$ , the number of processing steps

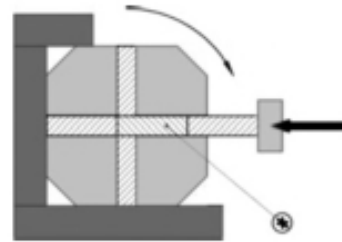
## Twist extrusion (TE)



$\varepsilon_{eff}^{min} \approx 0.4 + 0.1 \tan \gamma$ ;  $\varepsilon_{eff}^{max} \approx N \frac{2}{\sqrt{3}} \tan \gamma$   
 $\gamma$  is the twist line slope;  $N$  is the number of passes.  
 Deformation is non-uniform

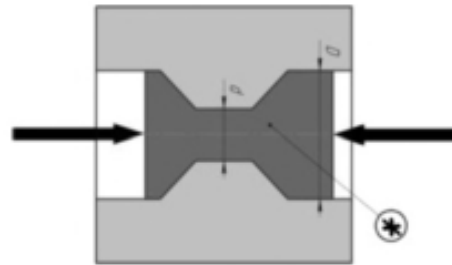
# Severe Plastic Deformation techniques

Rotary-die ECAP



ECAP equivalent

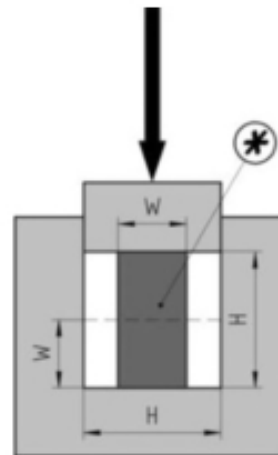
Cyclic extrusion-compression (CEC)



$$\epsilon_{eff} = N 4 \ln\left(\frac{D}{d}\right)$$

$N$ , number of cycles

Cyclic close-die forging (CCDF)



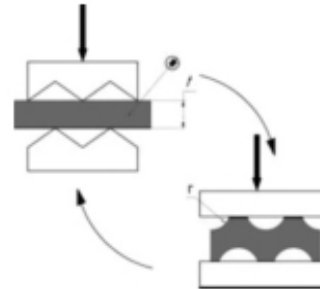
$$\epsilon_{eff} = N \frac{2}{\sqrt{3}} \ln\left(\frac{H}{W}\right)$$

$N$ , number of cycles

# Severe Plastic Deformation techniques



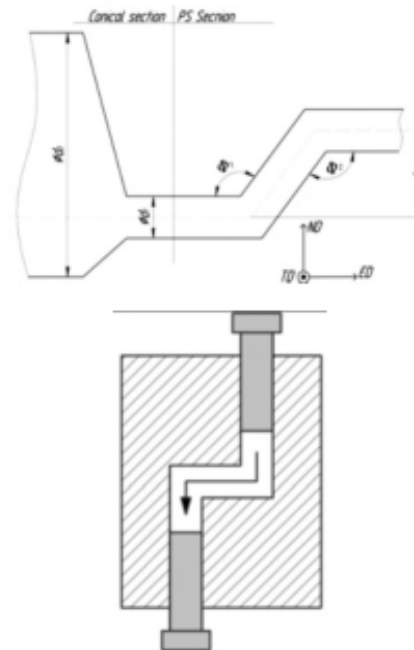
) Repetitive corrugation and straightening (RCS)



$$\varepsilon_{eff} = N \frac{4}{\sqrt{3}} \ln\left(\frac{r+t}{r+0.5t}\right)$$

$N$ , number of cycles

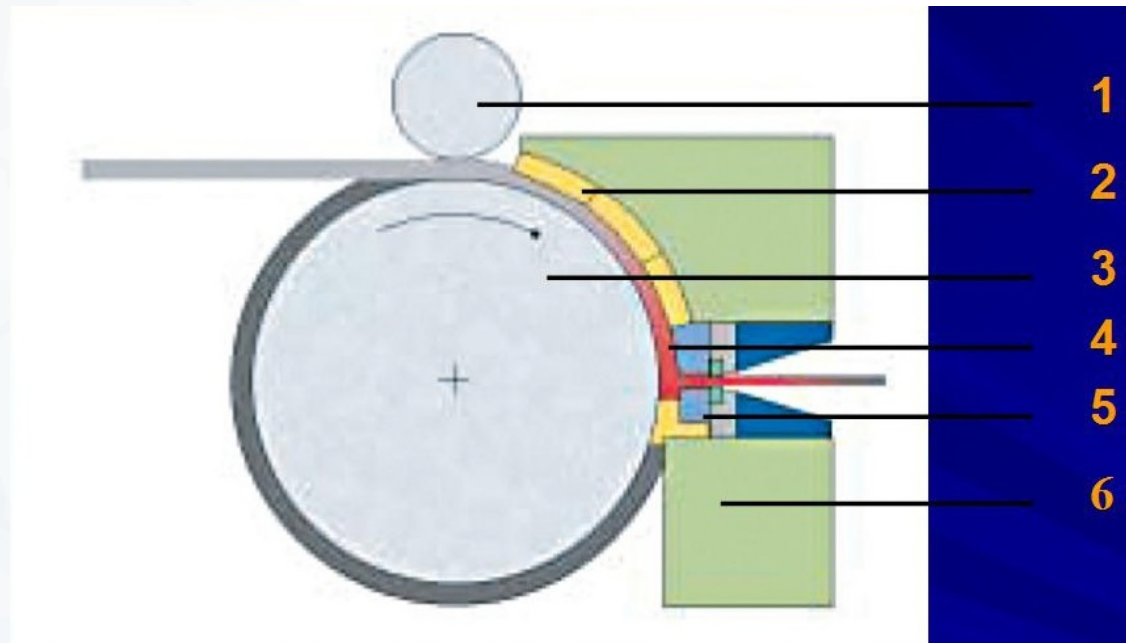
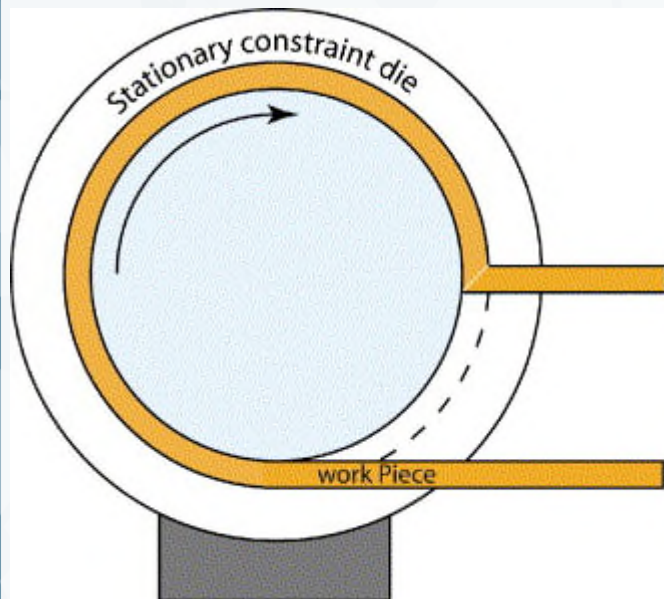
Integrated extrusion + ECAP



ECAP equivalent

## Continuous processes (industrial scale up)

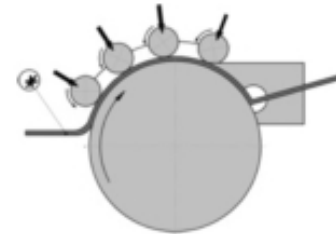
ECAP conform



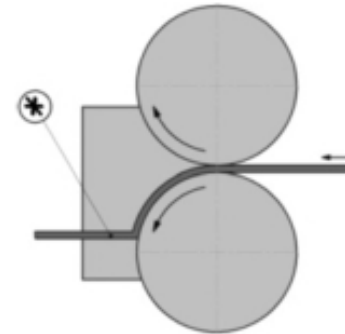
# Severe Plastic Deformation techniques



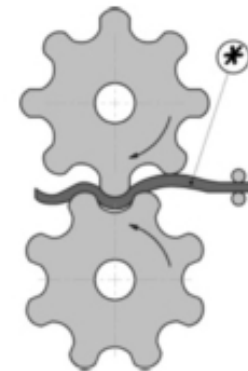
Con-shearing



Continuous confined strip shearing (C2S2)

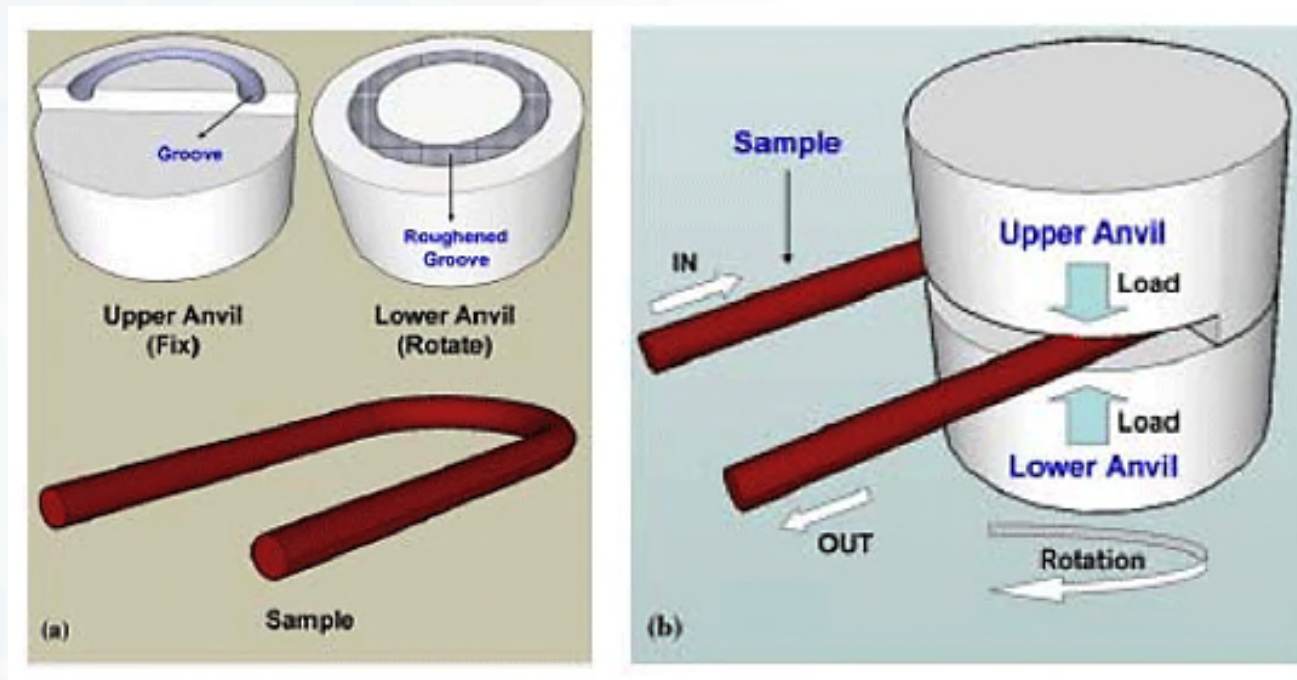


Continuous repetitive corrugating and straightening (RCS)



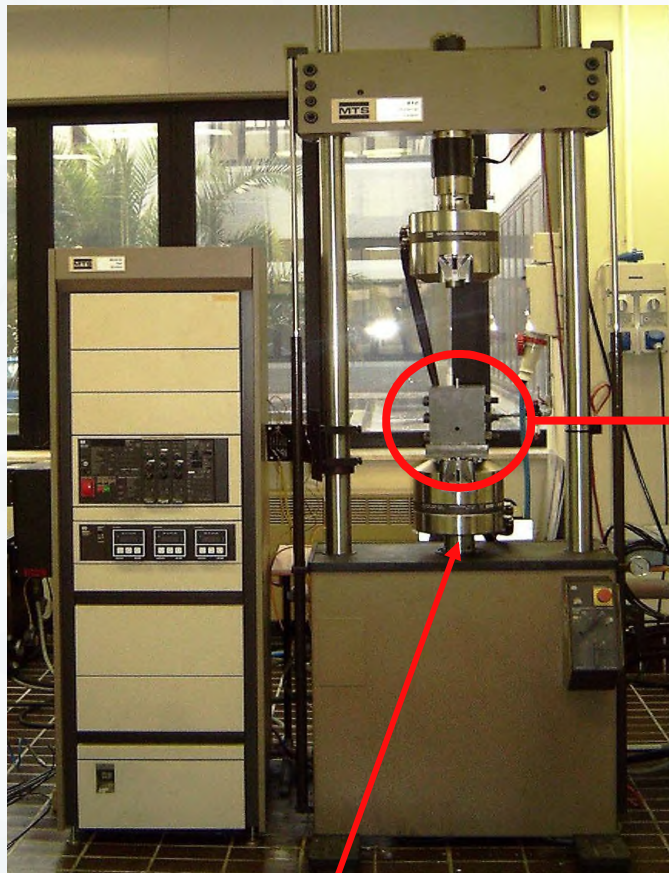
## Continuous processes (industrial scale up)

### Continuous HPT

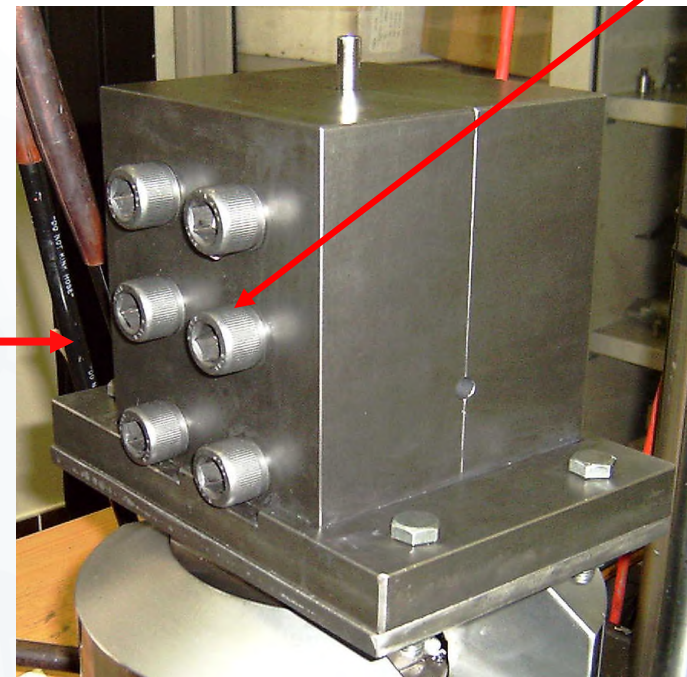


# Equal-Channel Angular Pressing (ECAP)

## Equal-Channel Angular Pressing (ECAP)



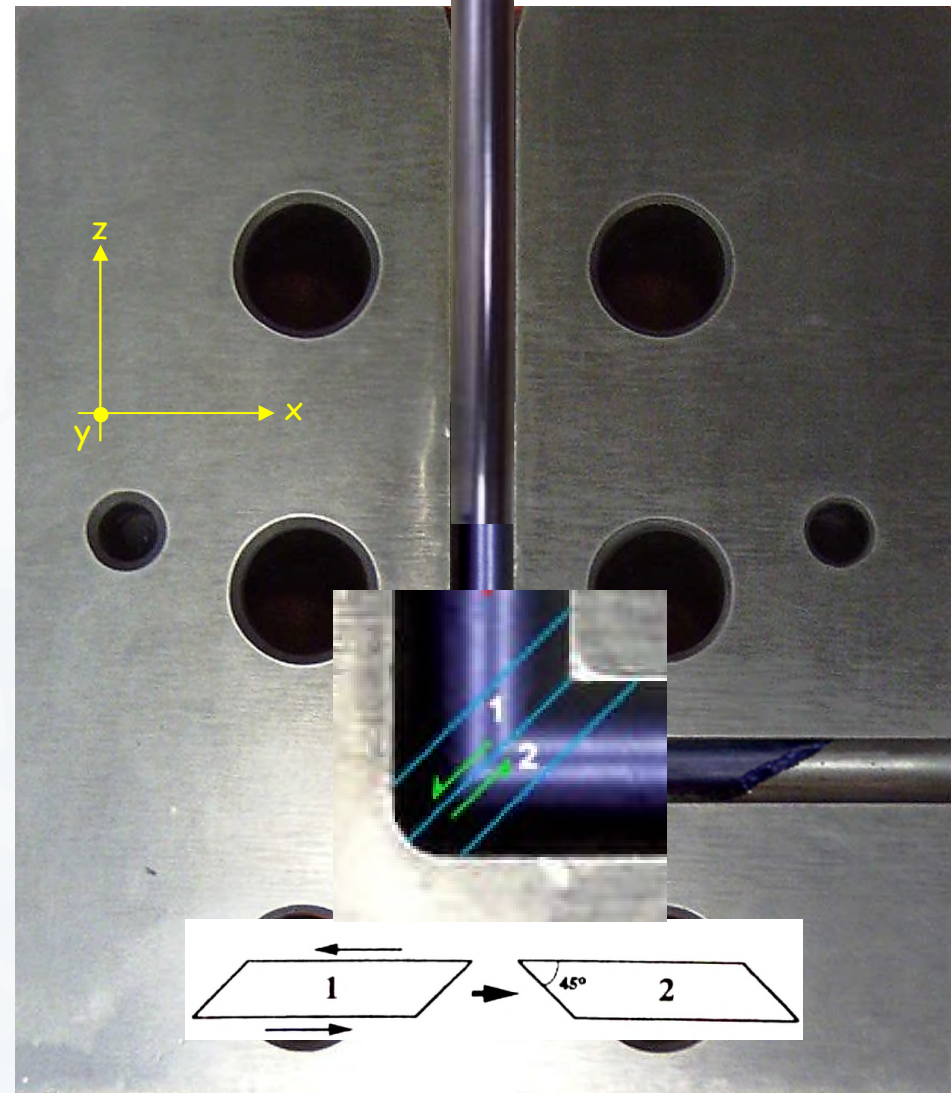
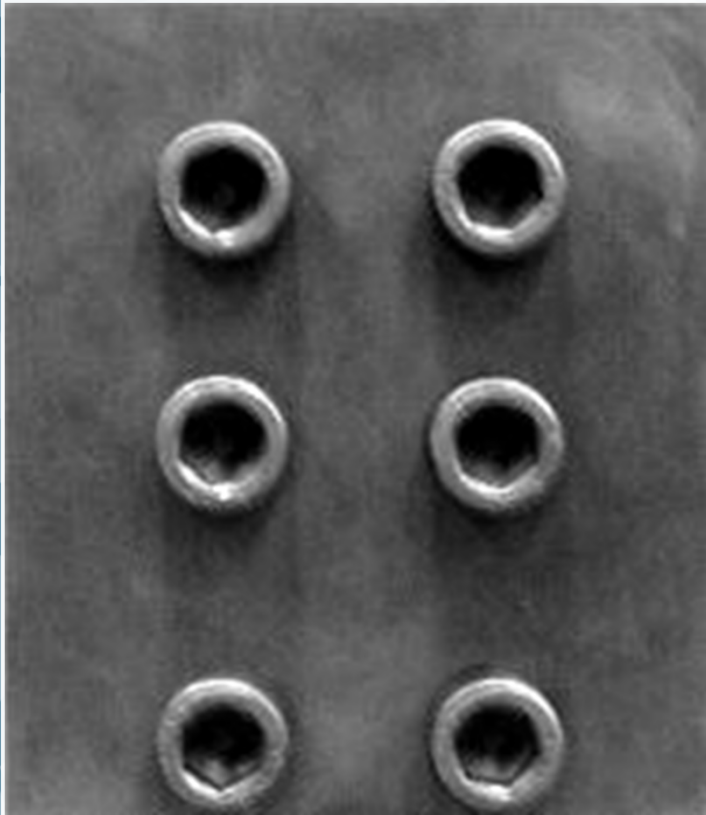
**250 kN max pressure**



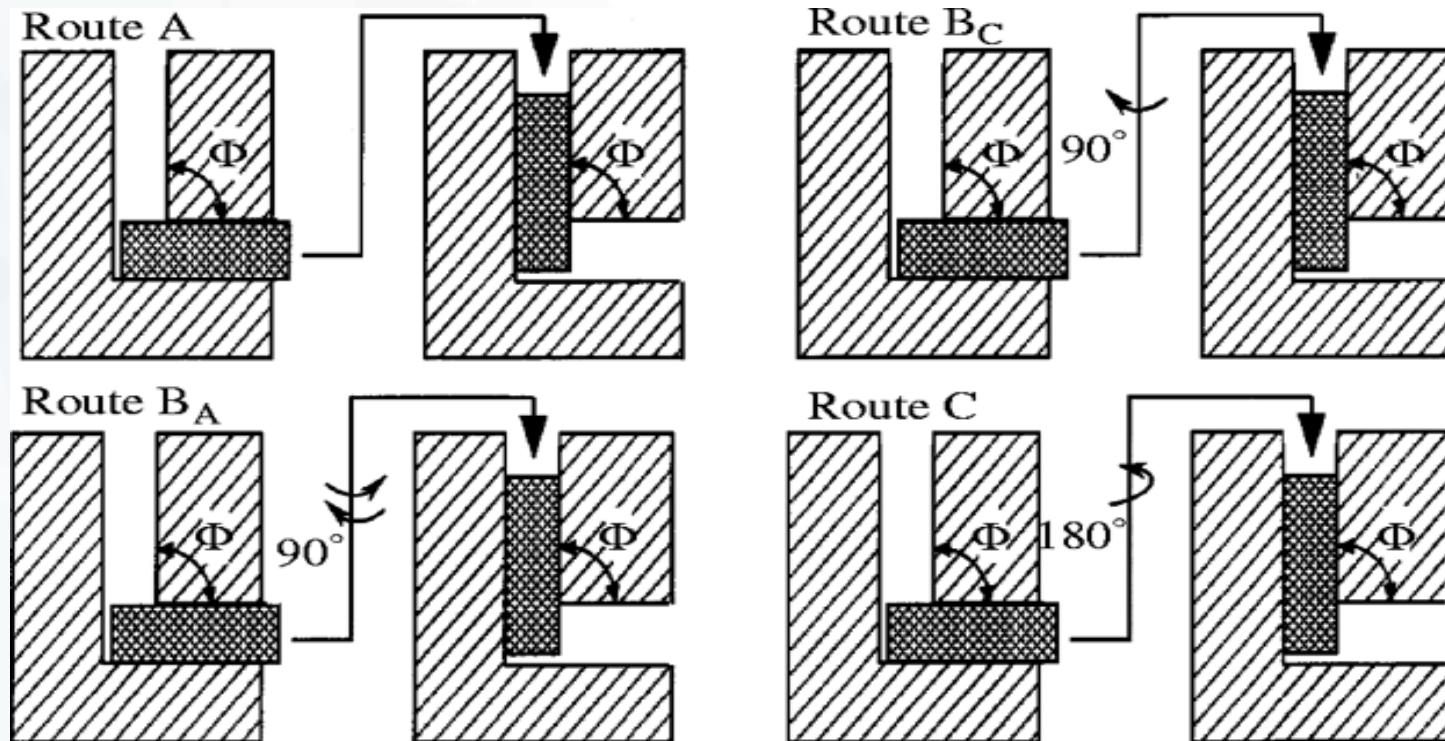
**6 long screws**

**the die**

# Equal-Channel Angular Pressing (ECAP)



## overview of routes scheme



After T.C. Langdon

Z. HORITA, T. FUJINAMI, M. NEMOTO AND T.G. LANGDON, Metall. Mater. Trans. A, 31A, (2000) p.691.

K. NAKASHIMA, Z. HORITA, M. NEMOTO AND T.G. LANGDON, Acta Mater., 46, 5, (1998) p.1589.

Y. IWAHASHI, M. FURAKAWA, Z. HORITA, M. NEMOTO AND T.G. LANGDON, Metall. Mater. Trans. A, 29A, (1998) p.2245.

M. FURUKAWA, Z. HORITA, M. NEMOTO, R.Z. VALIEV AND T.G. LANGDON, Acta Mater., 44, 11, (1996) p.4619.

# Equal-Channel Angular Pressing (ECAP)



## route A

			Number of pressings				
			1	2	3	4	5
90° X A	X						
	Y						
	Z						

## route Bc

			Number of pressings				
			1	2	3	4	5
90° X Bc	X						
	Y						
	Z						

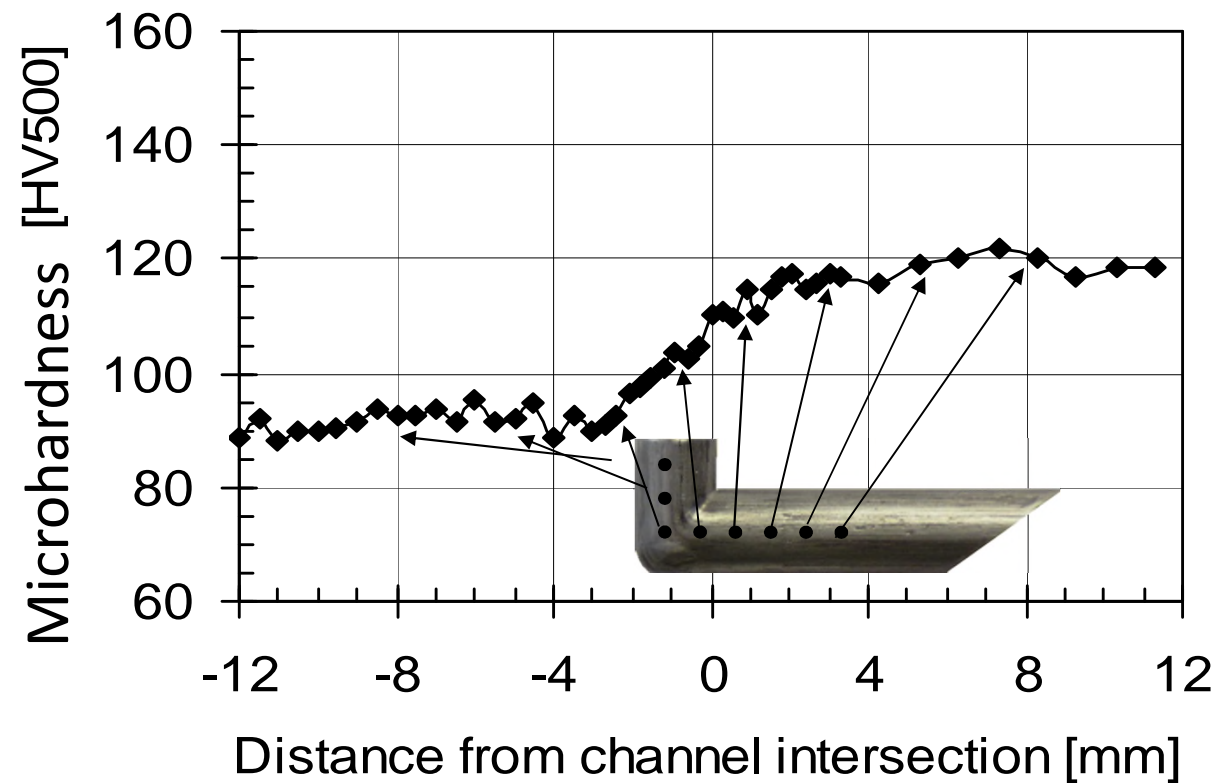
## route BA

			Number of pressings				
			1	2	3	4	5
90° X BA	X						
	Y						
	Z						

## route C

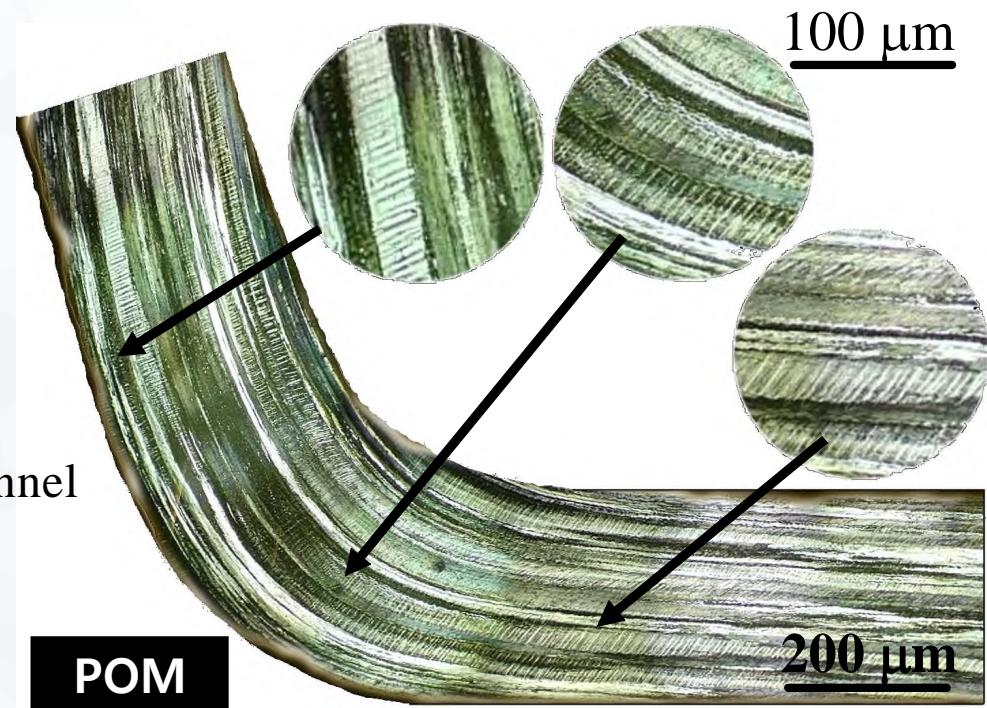
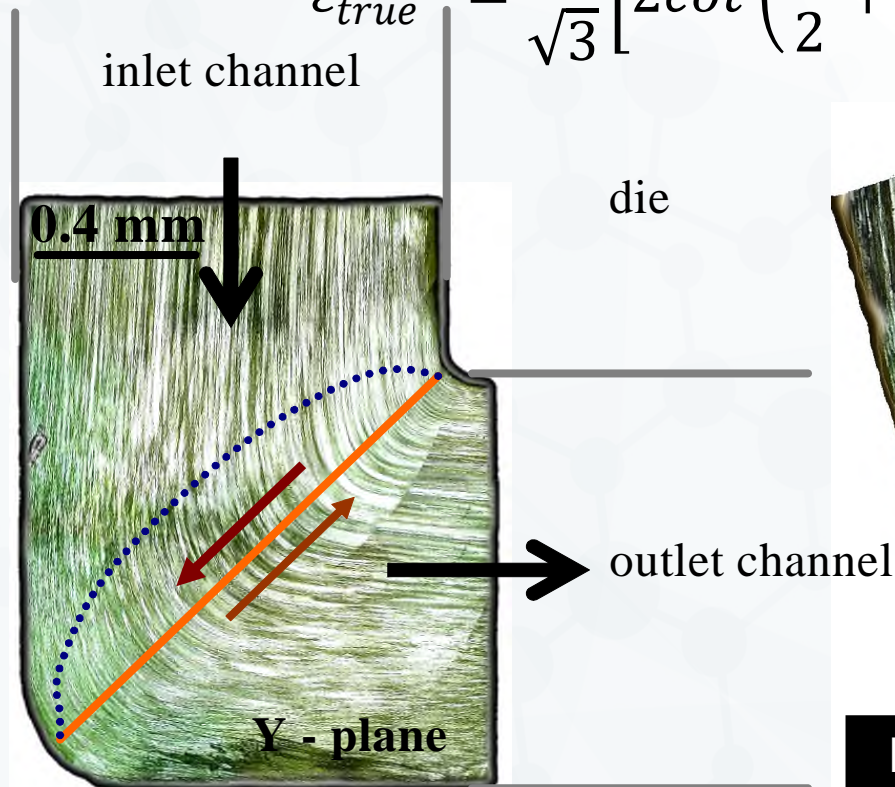
			Number of pressings				
			1	2	3	4	5
90° X C	X						
	Y						
	Z						

Typical microhardness profile along sample longitudinal axis  
at the passage into the ECAP die



## ECAP induced grain modifications

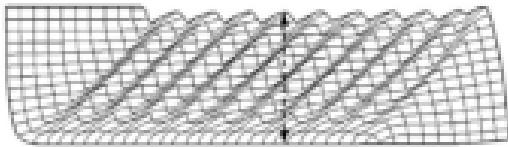
$$\varepsilon_{true}^{ECAP} = \frac{N}{\sqrt{3}} \left[ 2 \cot \left( \frac{\Phi}{2} + \frac{\Psi}{2} \right) + \Psi \operatorname{cosec} \left( \frac{\Phi}{2} + \frac{\Psi}{2} \right) \right]$$



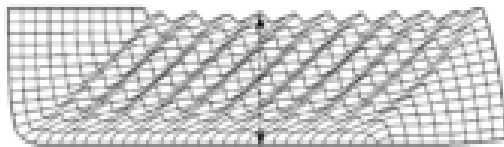
die

FEM – Transition from unstable to stable flow

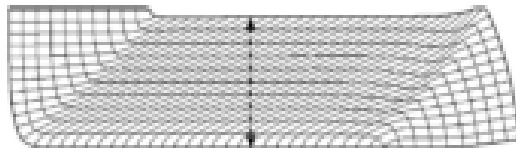
$m = 0$



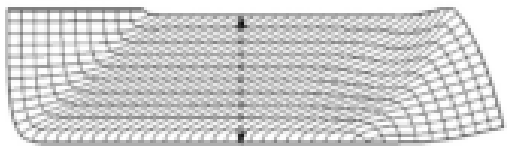
$m = 0.01$



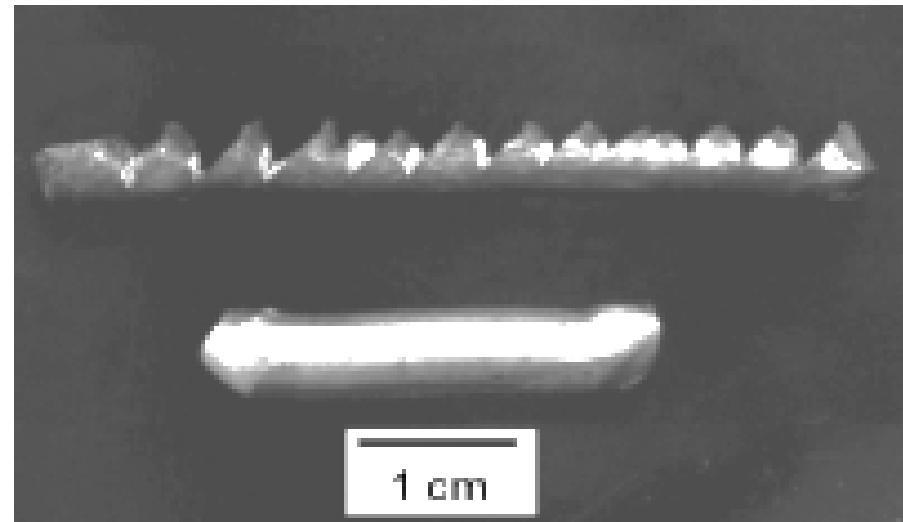
$m = 0.05$



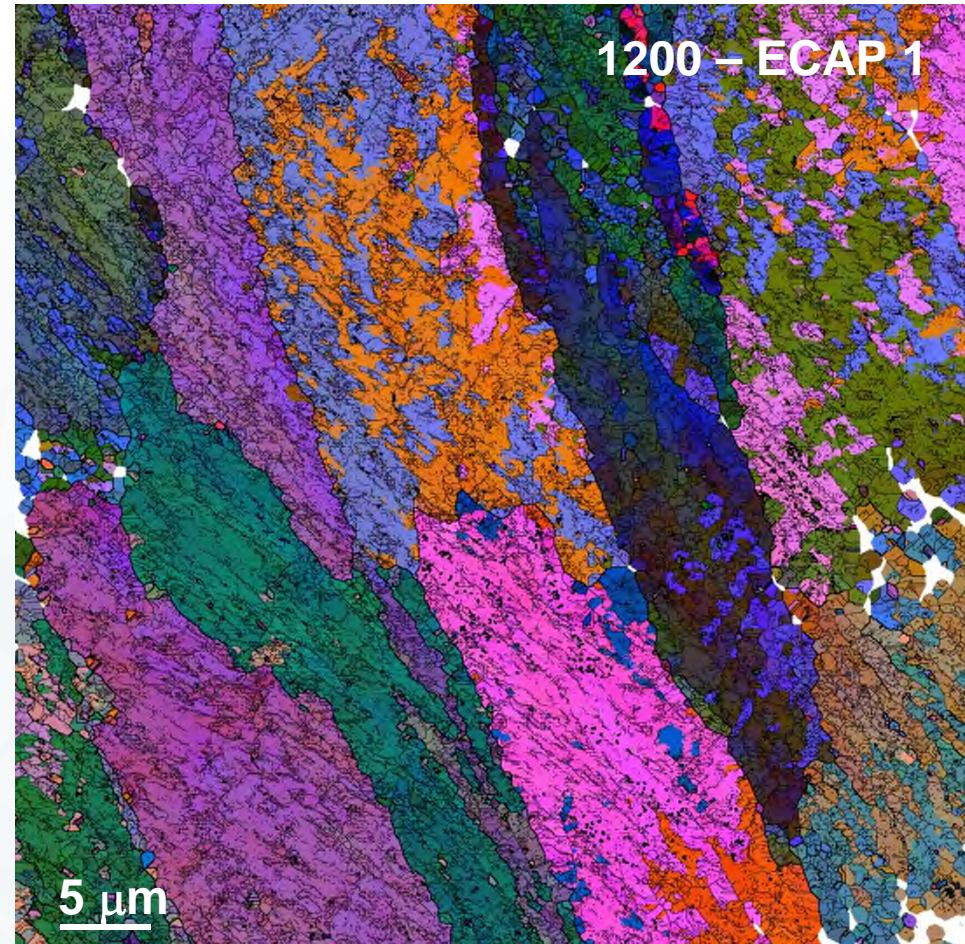
$m = 0.1$



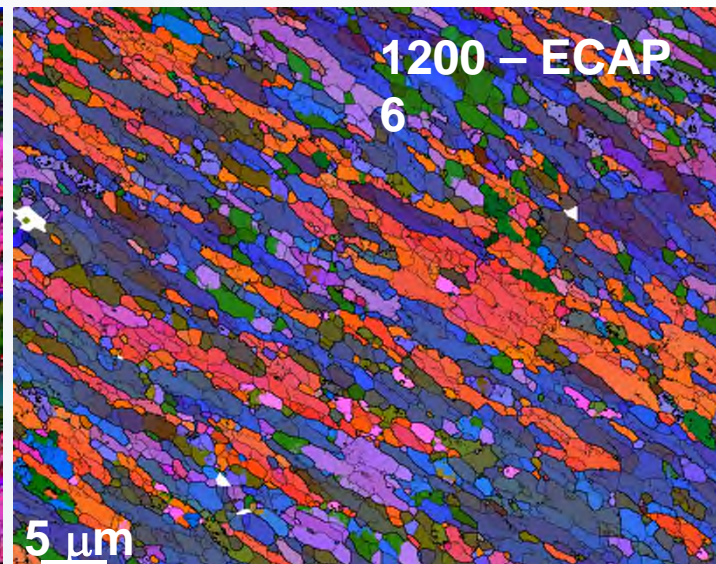
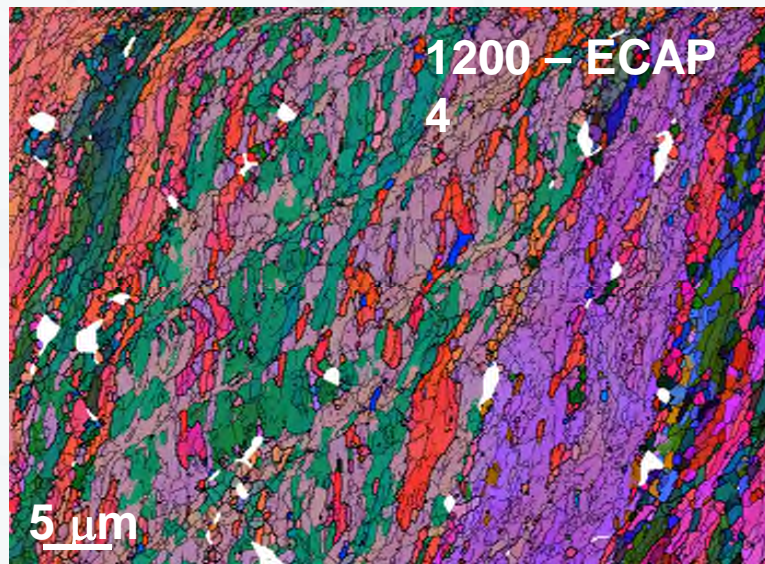
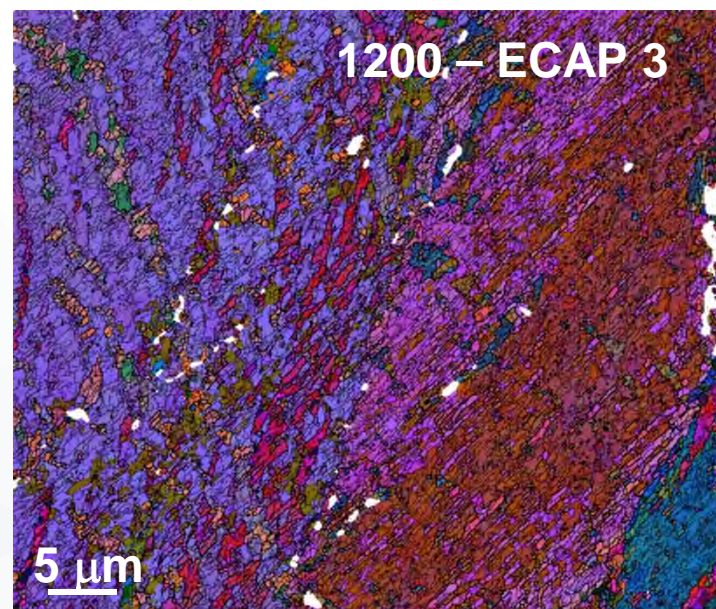
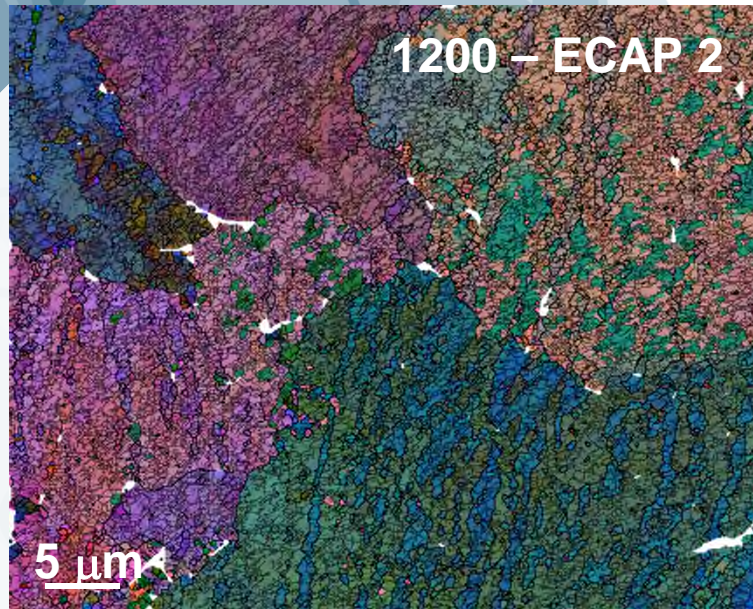
## Sample and set-up



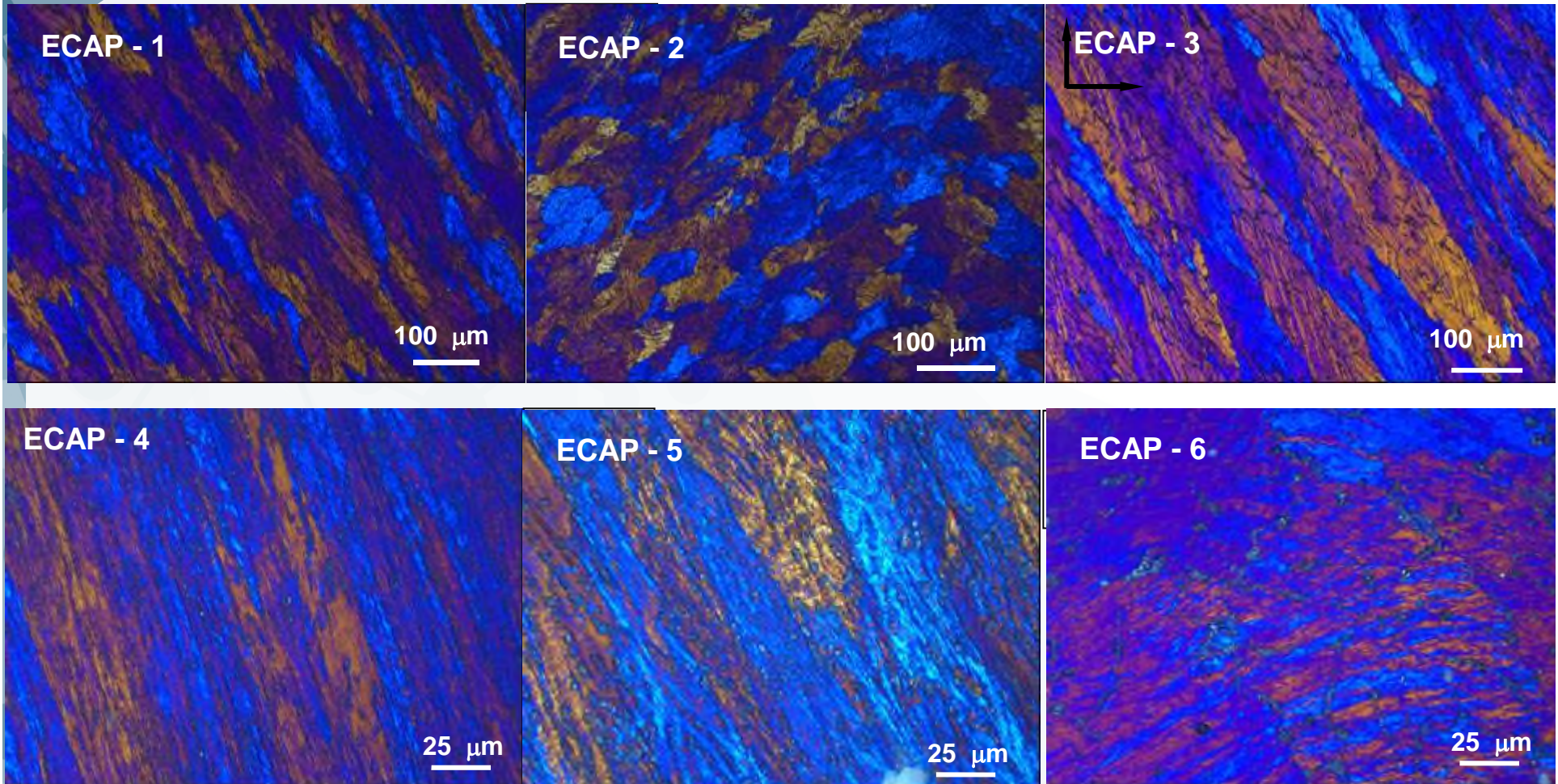
# Equal-Channel Angular Pressing (ECAP)



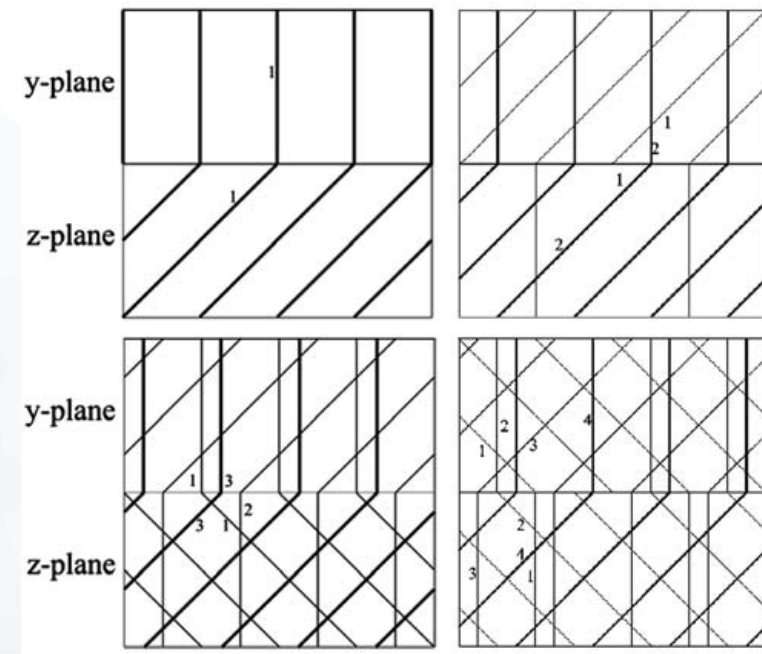
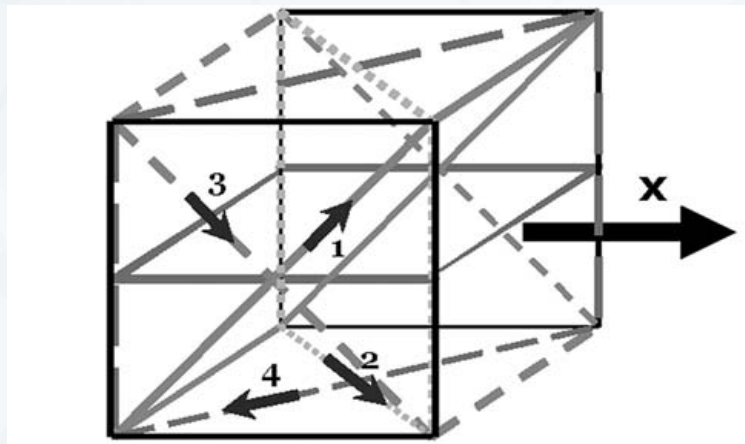
# Equal-Channel Angular Pressing (ECAP)



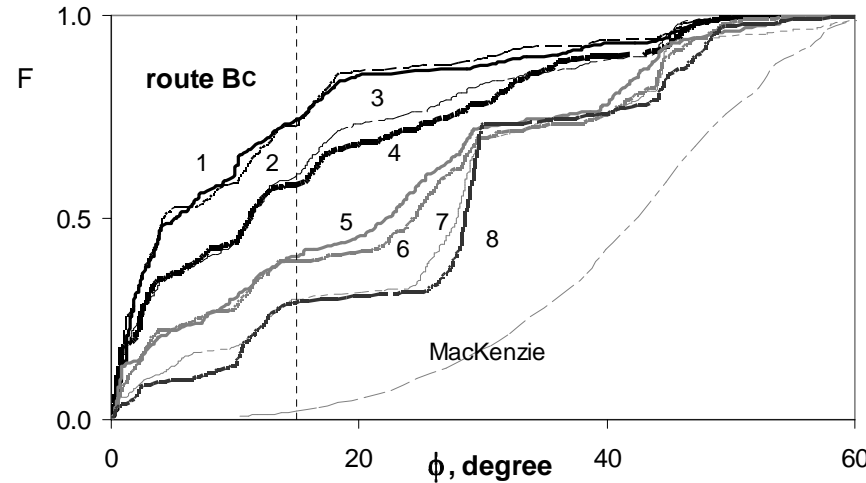
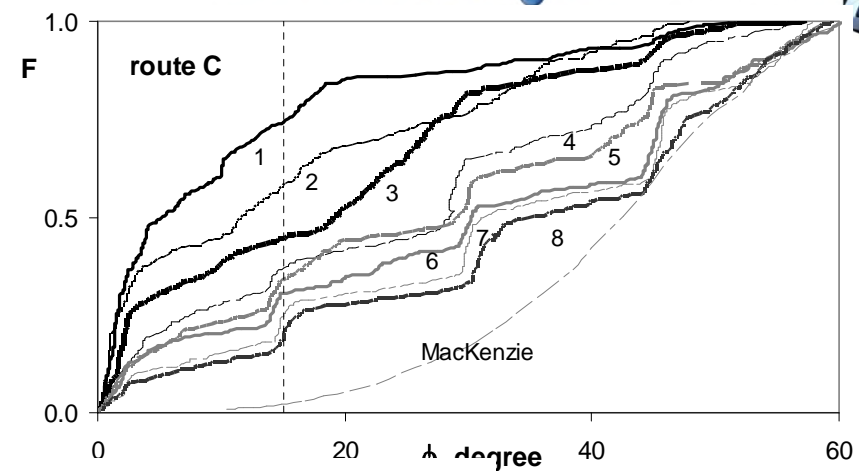
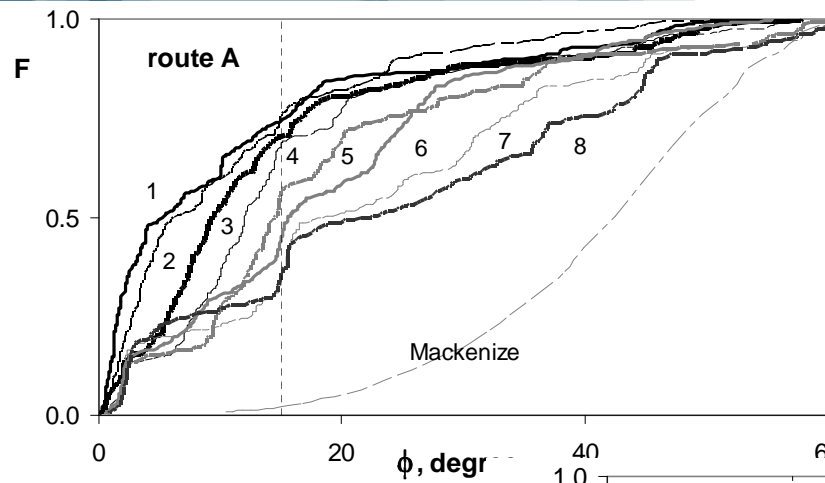
# Equal-Channel Angular Pressing (ECAP)



## Shear plane deformation activated during ECAP Route Bc



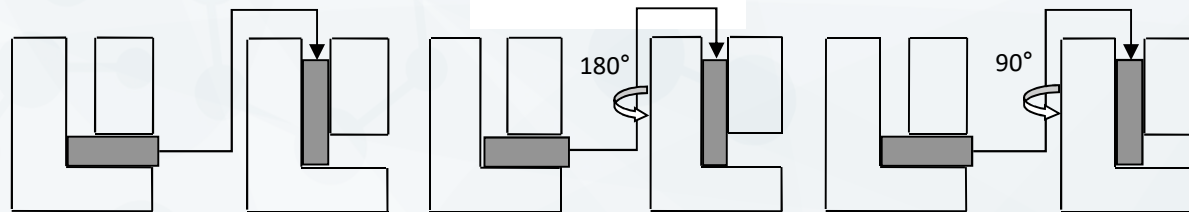
# Equal-Channel Angular Pressing (ECAP)



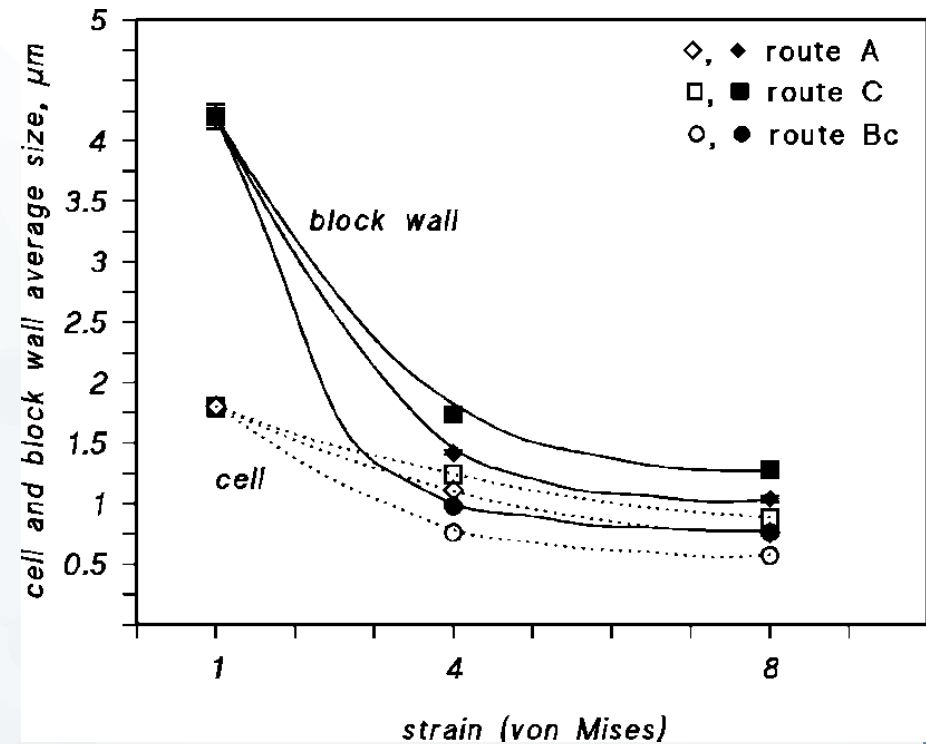
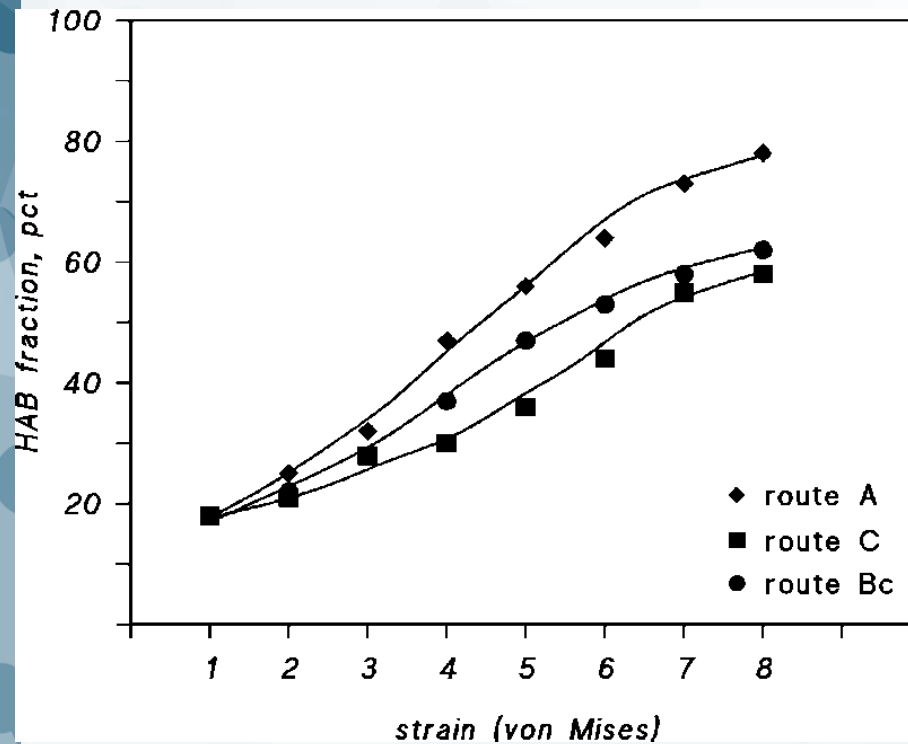
Route A

Route C

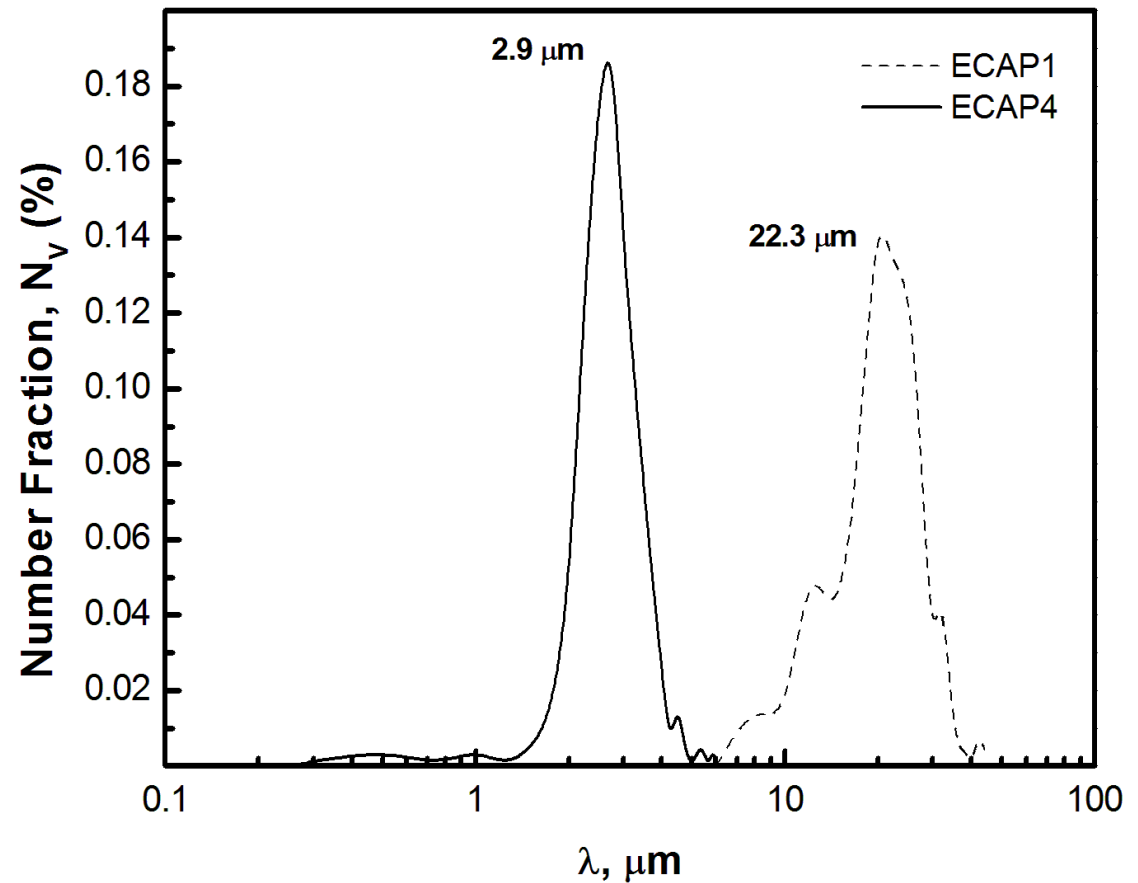
Route B<sub>c</sub>



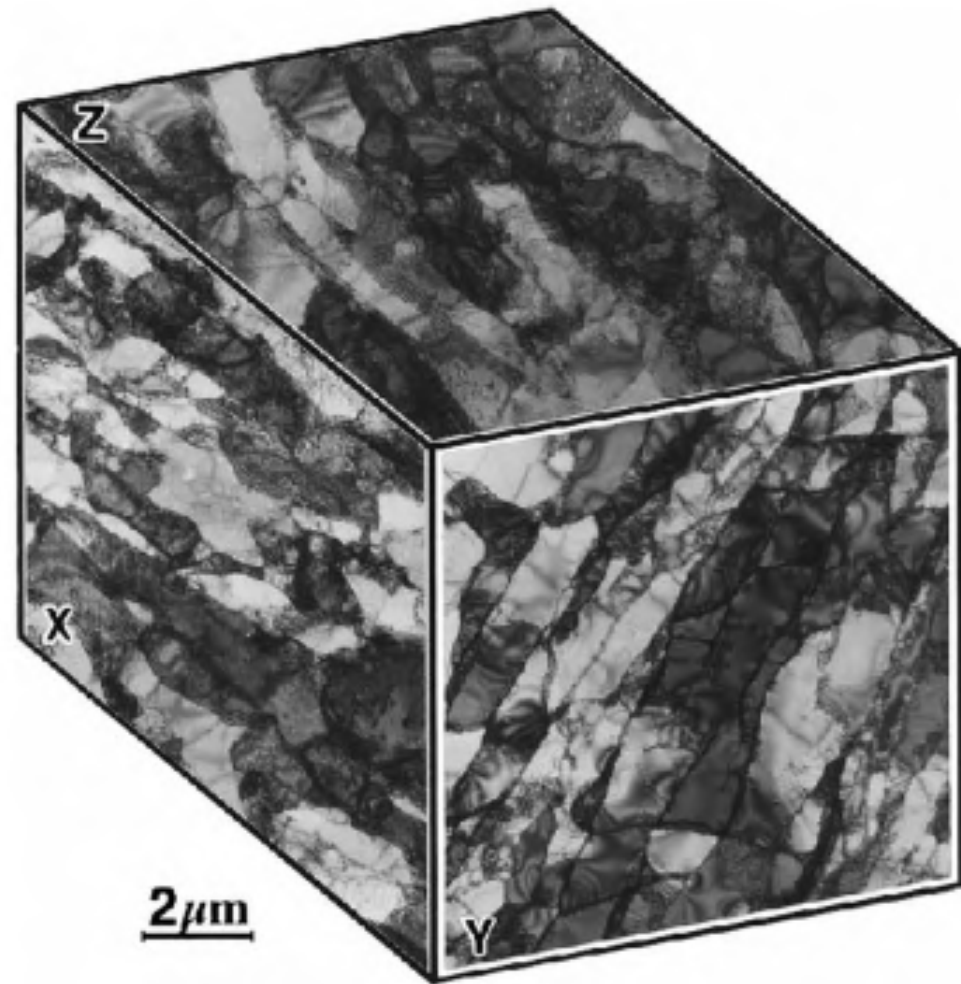
## Grain size evolution



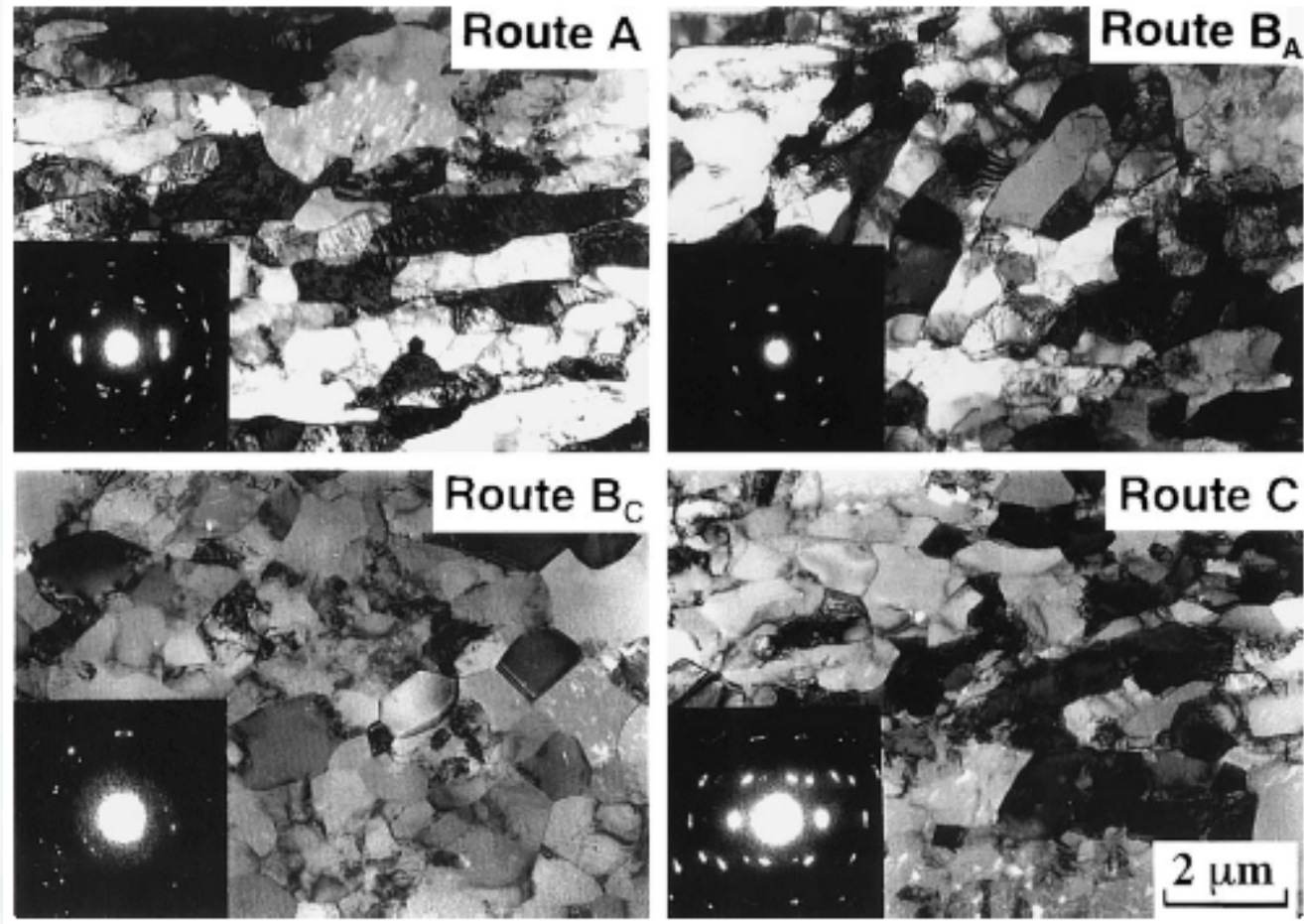
## Grain size evolution



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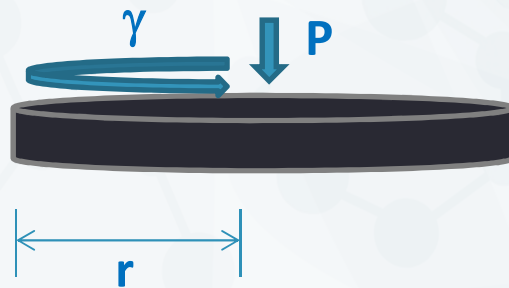
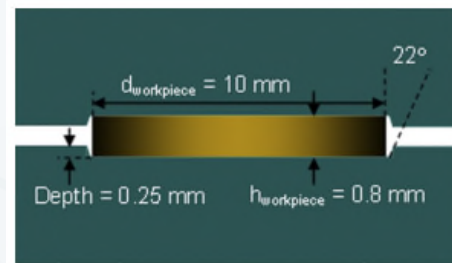
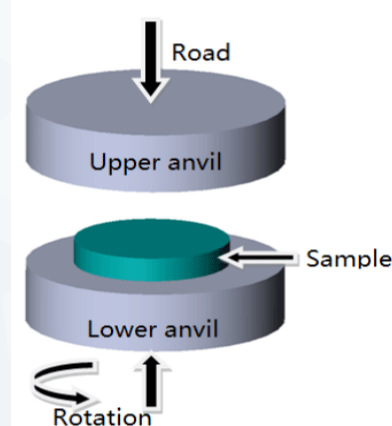
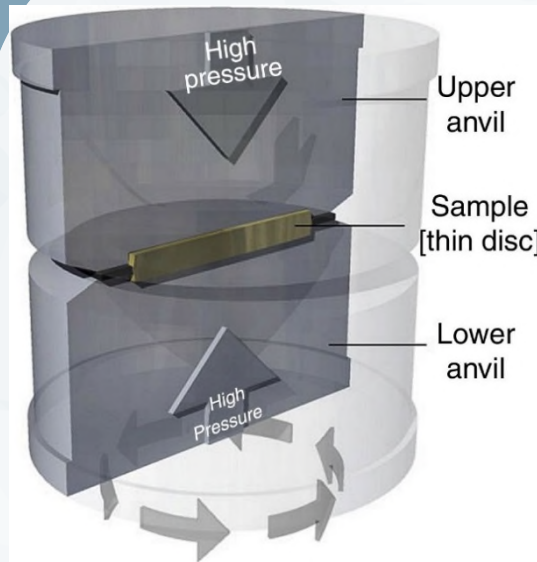


## Grain size evolution



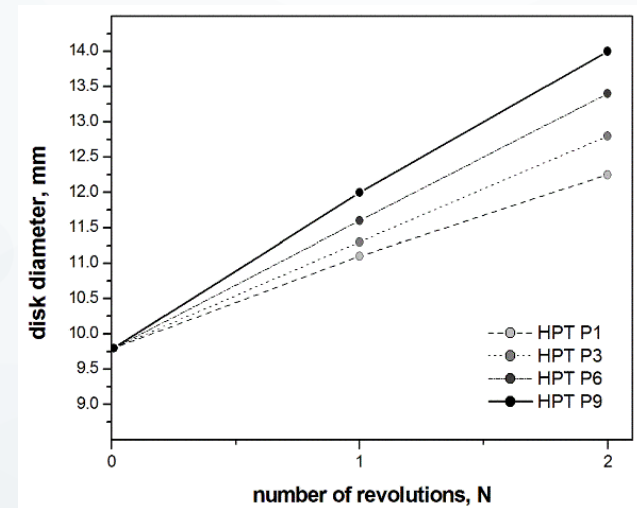
# High-Pressure Torsion (HPT)

# High-Pressure Torsion (HPT)



$$t = f(\gamma, P) \quad d\gamma = r \cdot d\theta / t$$

$$\epsilon_{\text{true}}^{\text{HPT}} = \ln \left( 1 + \left( \frac{\theta r}{t} \right) \right) + \ln \left( \frac{t_0}{t} \right)$$



# High-Pressure Torsion (HPT)

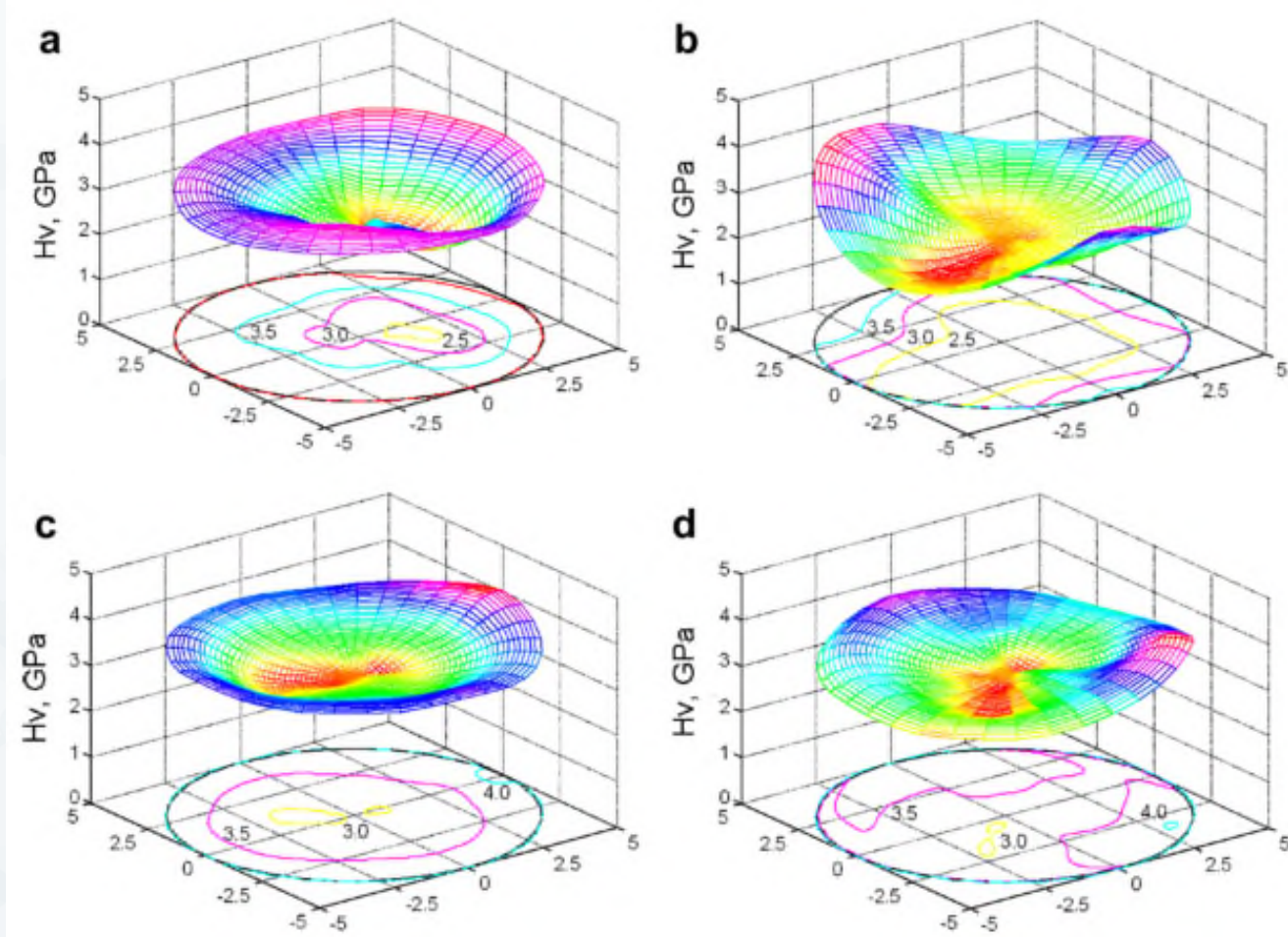
## HPT

P = 1 GPa (a);

P = 3 GPa (b);

P = 6 GPa (c);

P = 9 GPa (d)



# High-Pressure Torsion (HPT)

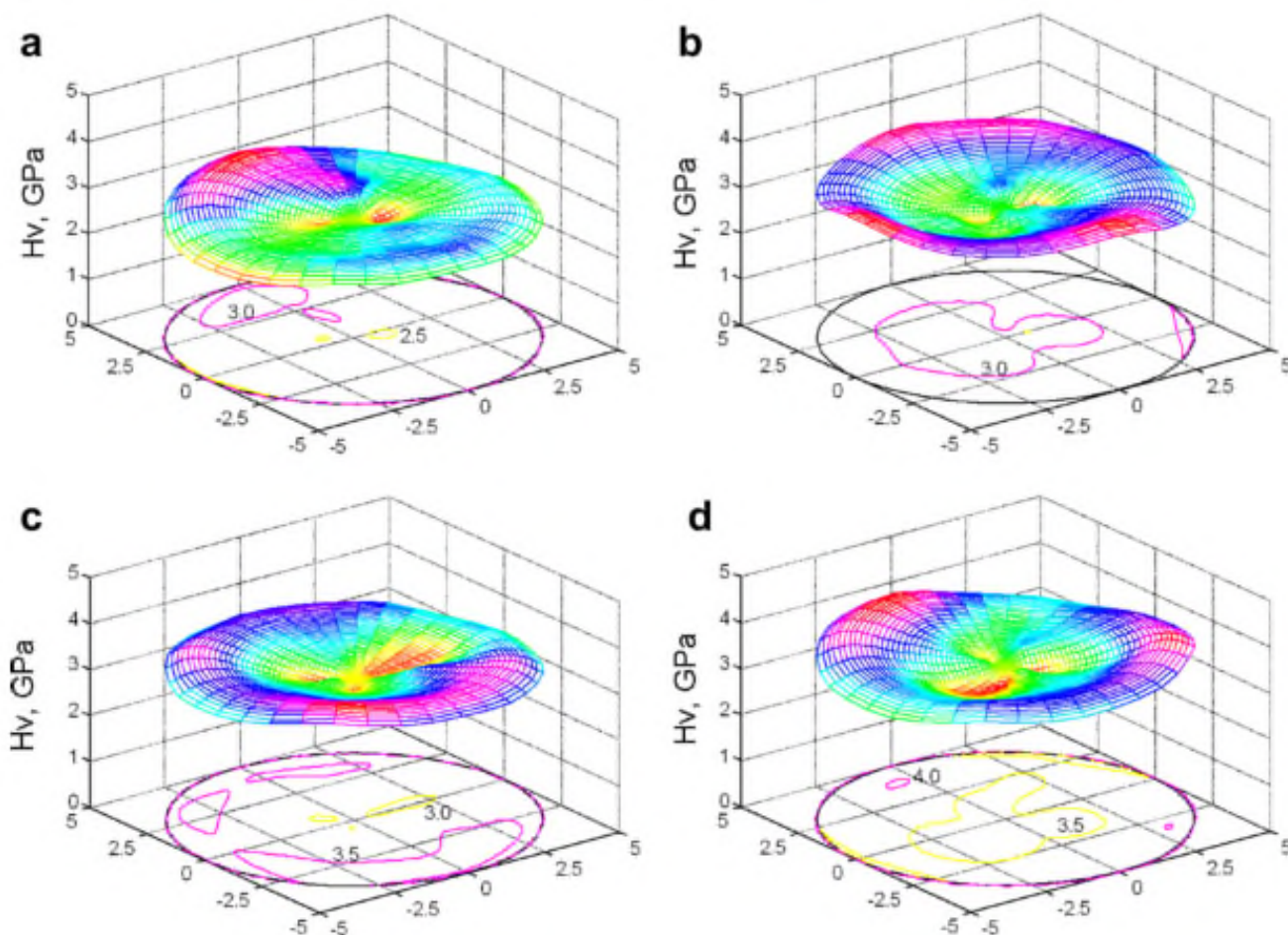
**HPT @ P=6GPa**

Turns = 0.5 (a);

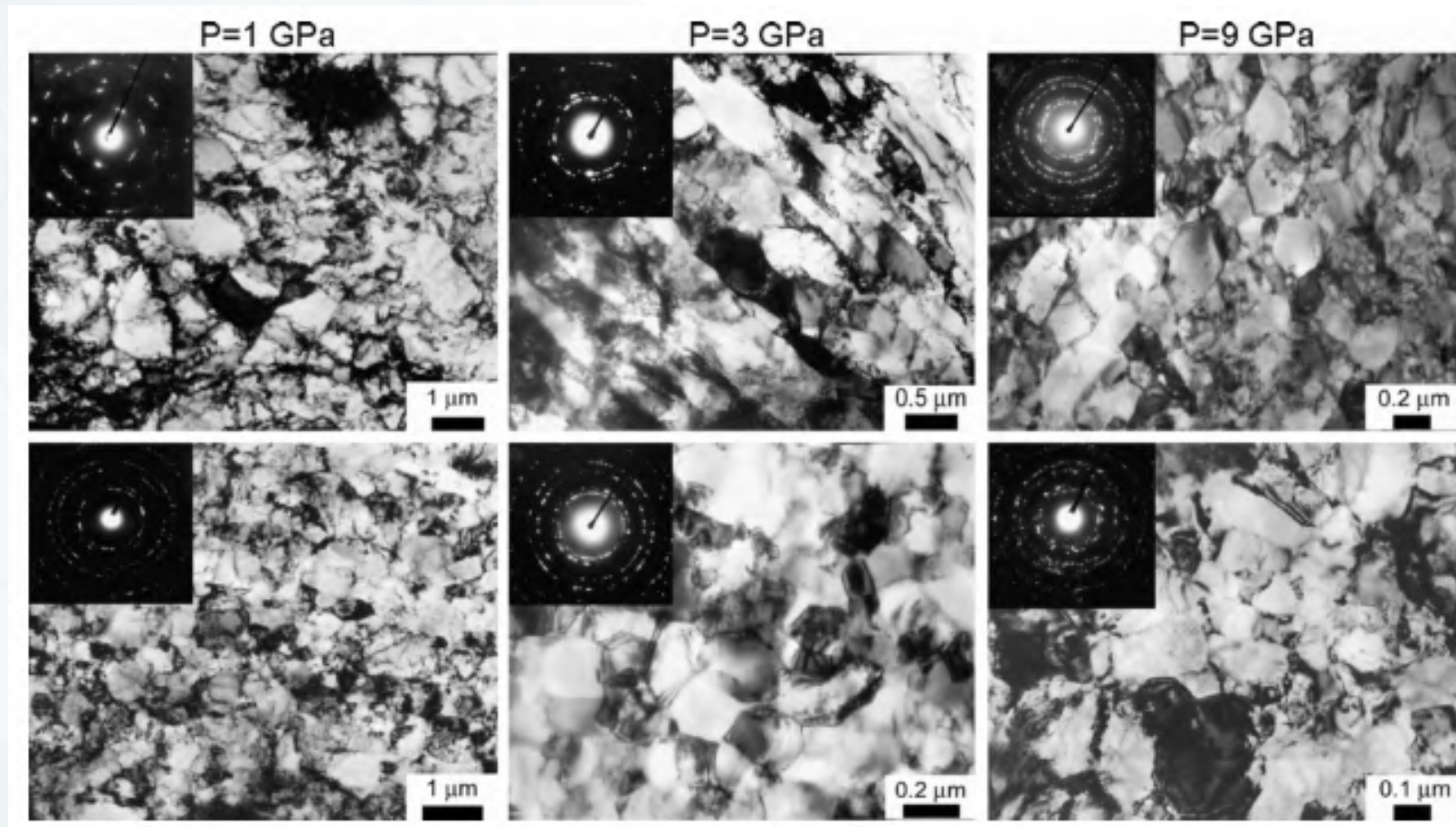
Turns = 1 (b);

Turns = 3 (c);

Turns = 7 (d)



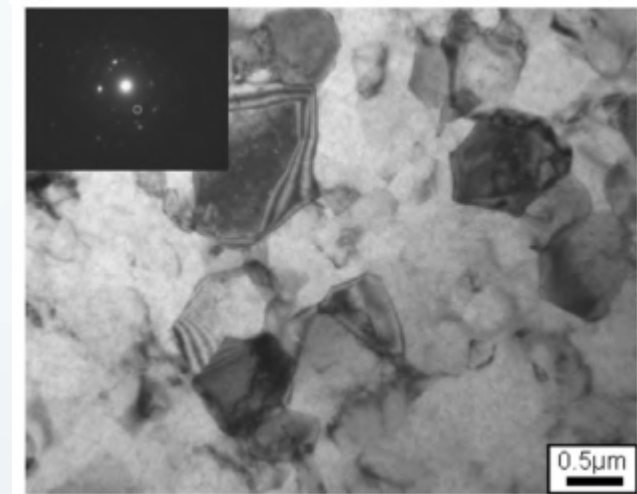
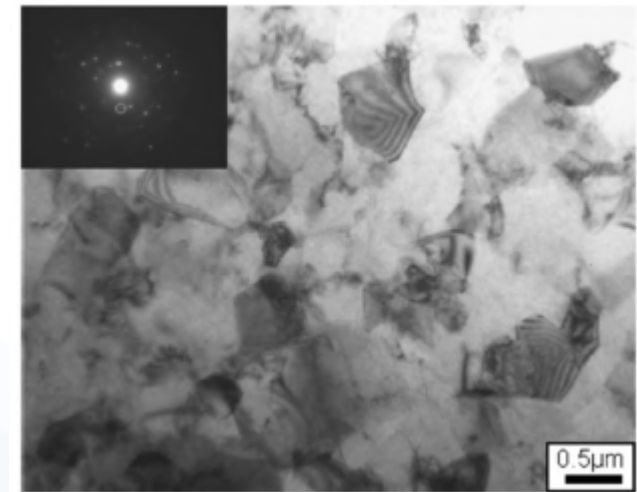
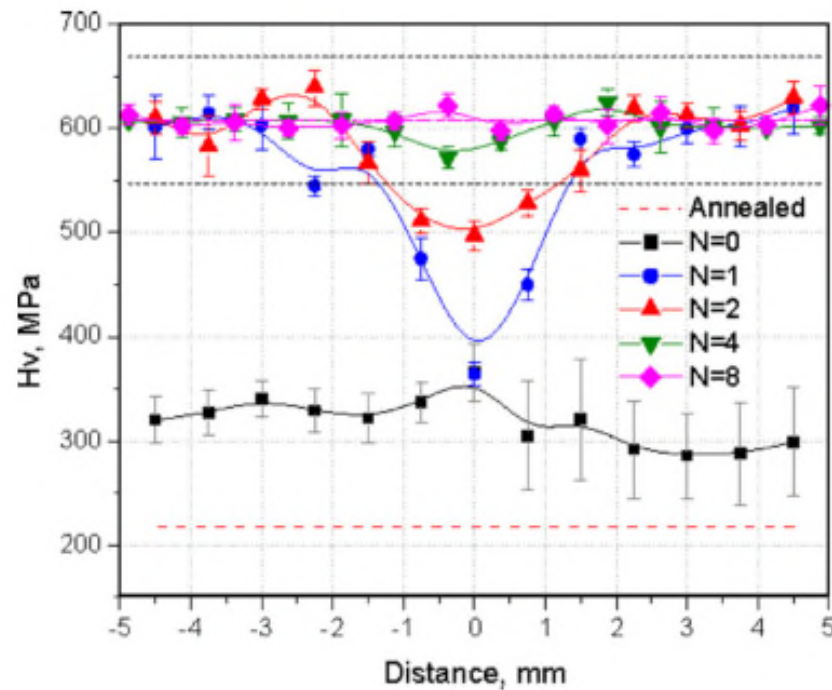
## Grain size evolution



# High-Pressure Torsion (HPT)

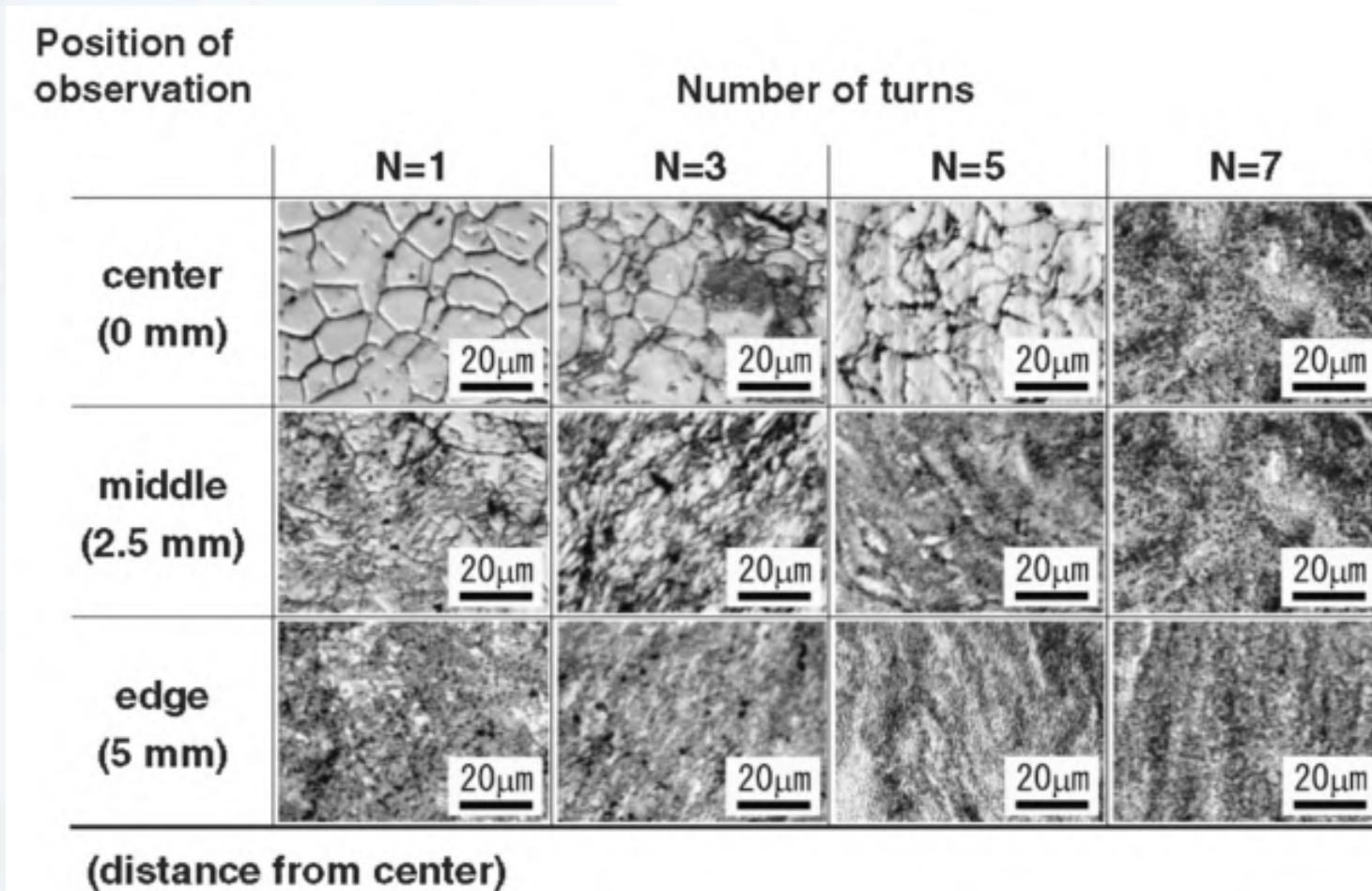
## Microstructure homogeneization with HPT straining

central region



near periphery

## Grain size evolution



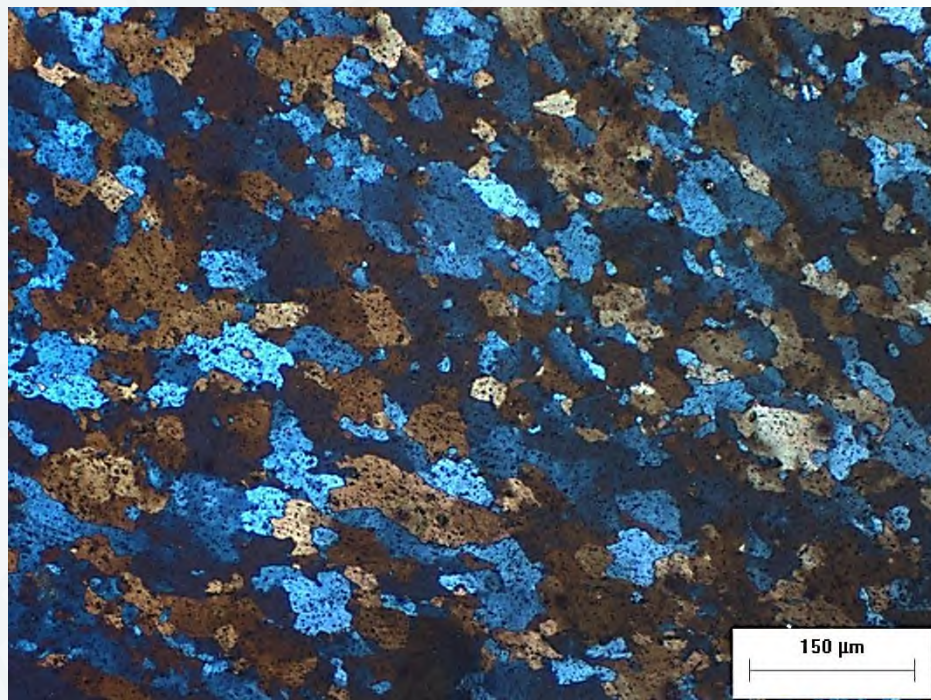
# Combined effect of ECAP and HPT

## Combined effect of ECAP + HPT



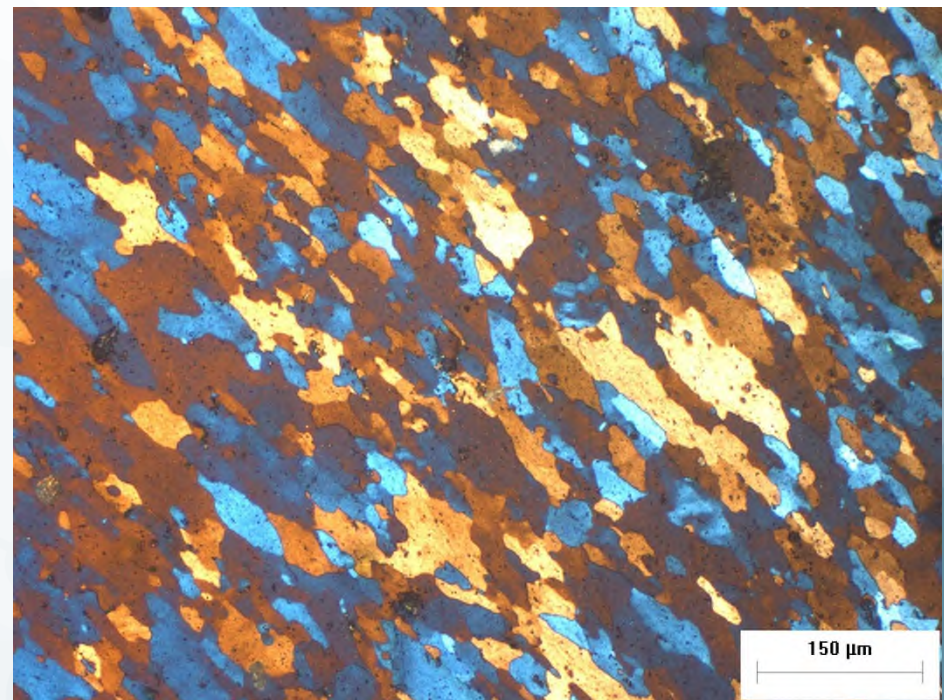
### ECAP-4 / HPT

P=9GPa – opposite rotation direction  
with respect previous ECAP shearing



### ECAP-4 / HPT

P=9GPa – concurrent rotation direction  
with respect previous ECAP shearing

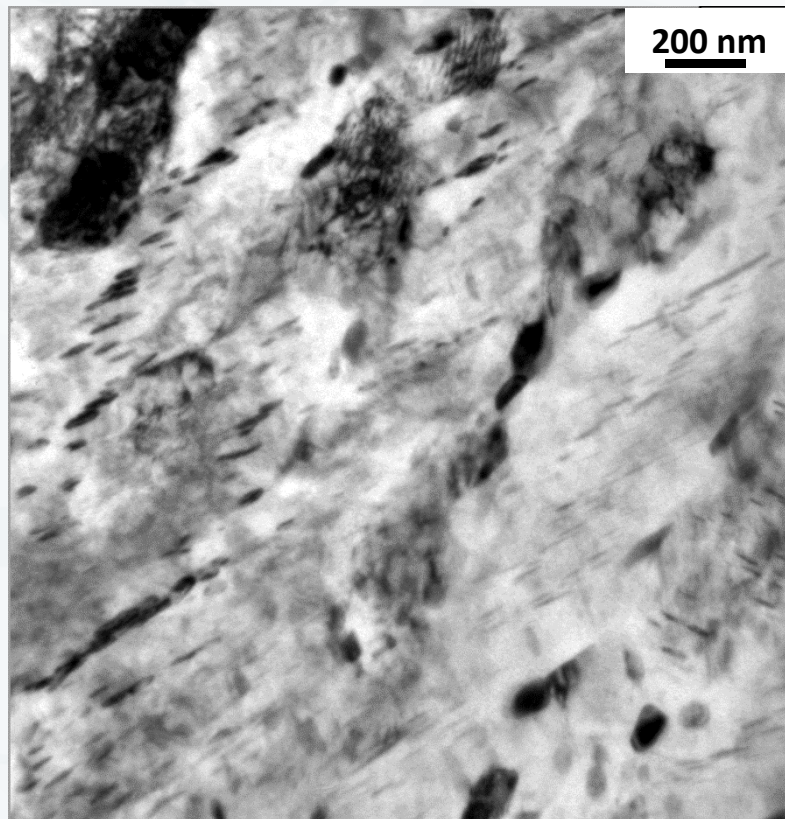


## Combined effect of ECAP + HPT



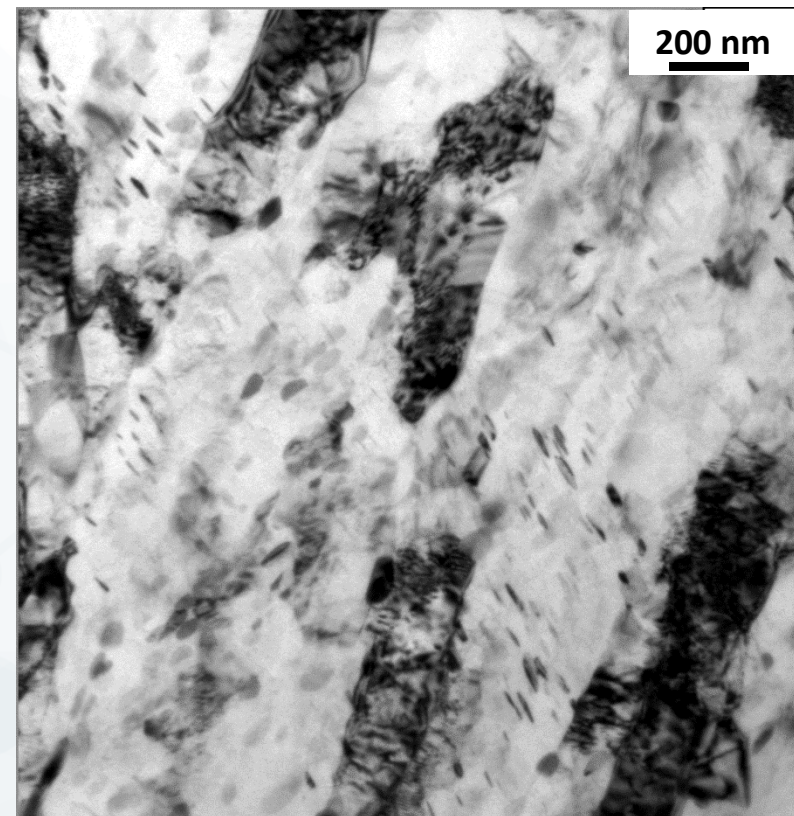
### ECAP-4 / HPT

P=9GPa – opposite rotation direction  
with respect previous ECAP shearing



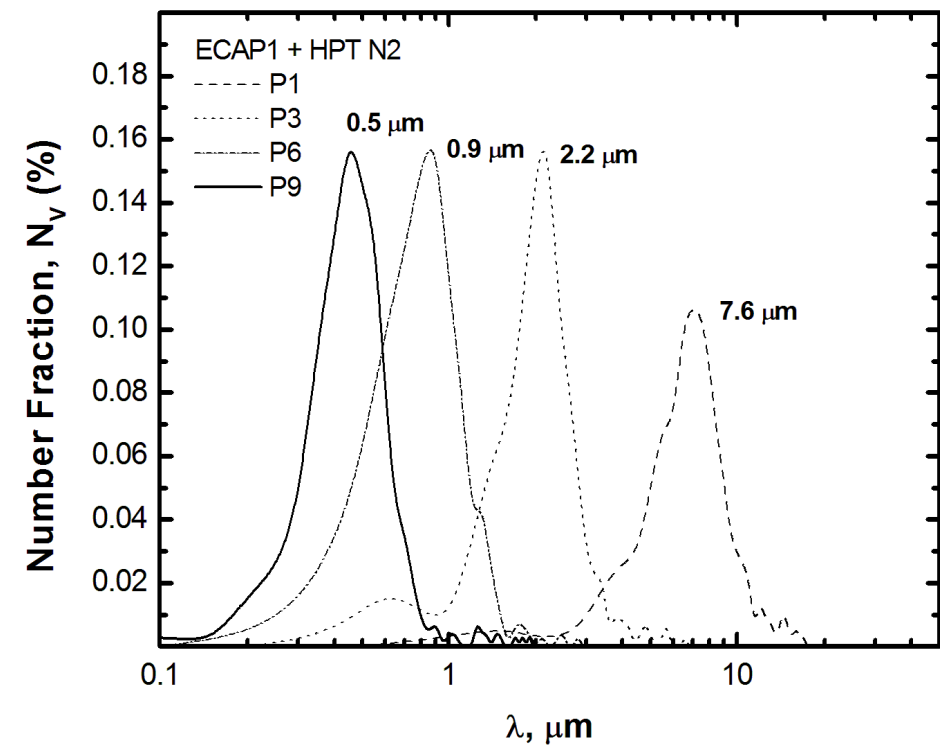
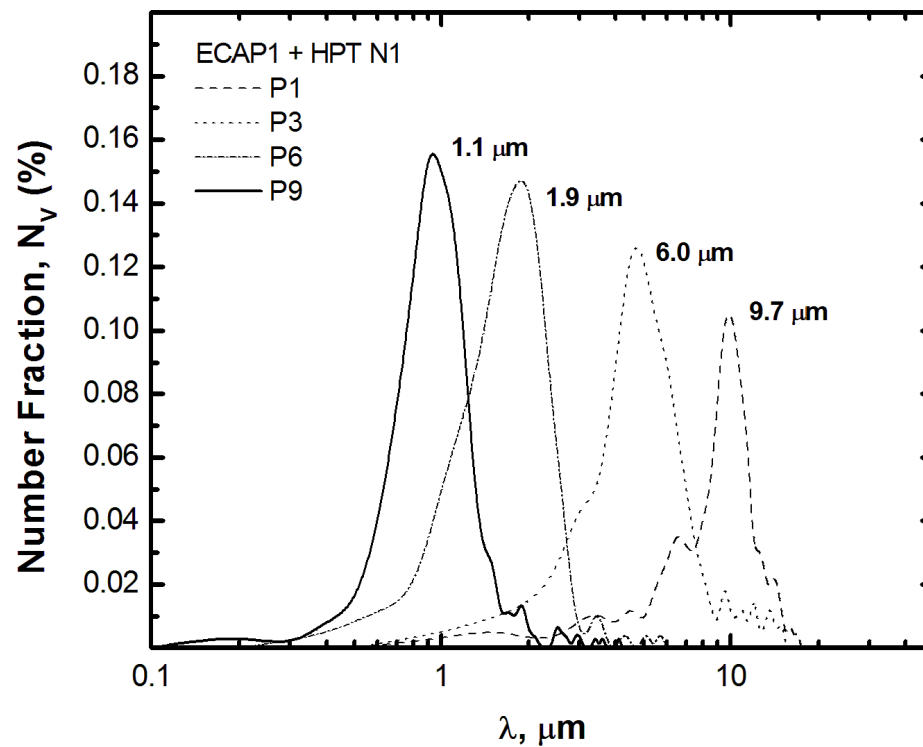
### ECAP-4 / HPT

P=9GPa – concurrent rotation direction  
with respect previous ECAP shearing



## Grain size evolution

ECAP-1 / HPT

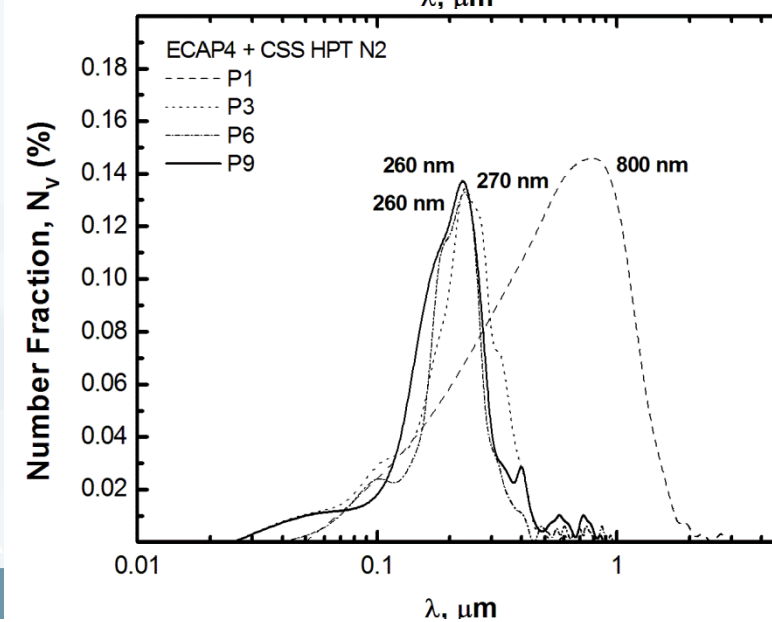
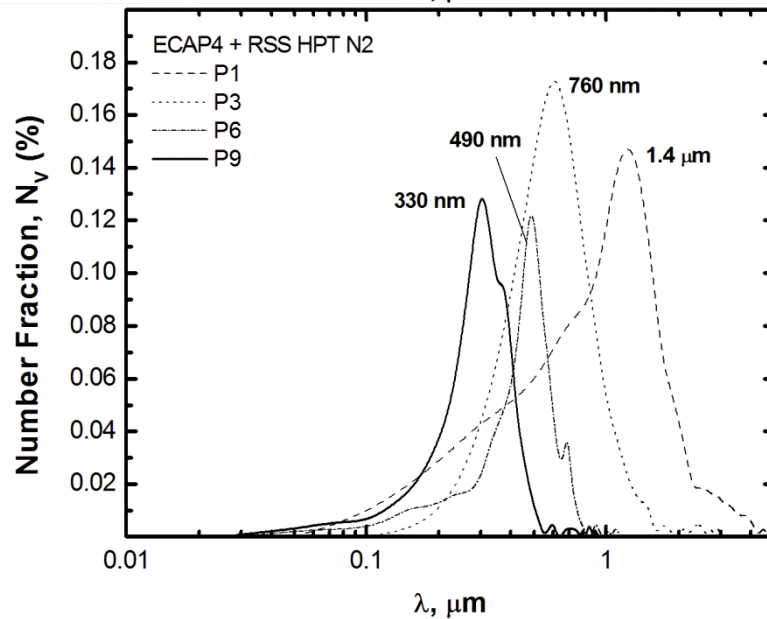
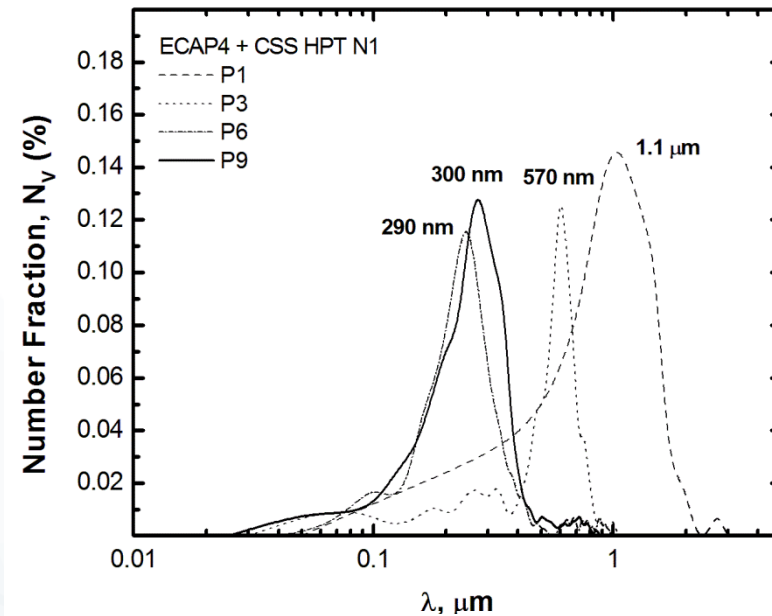
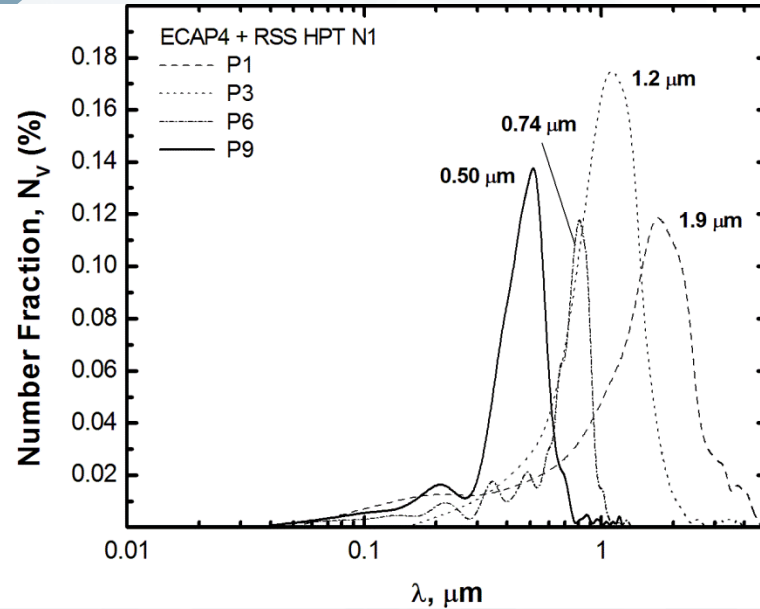


# Combined effect of ECAP + HPT

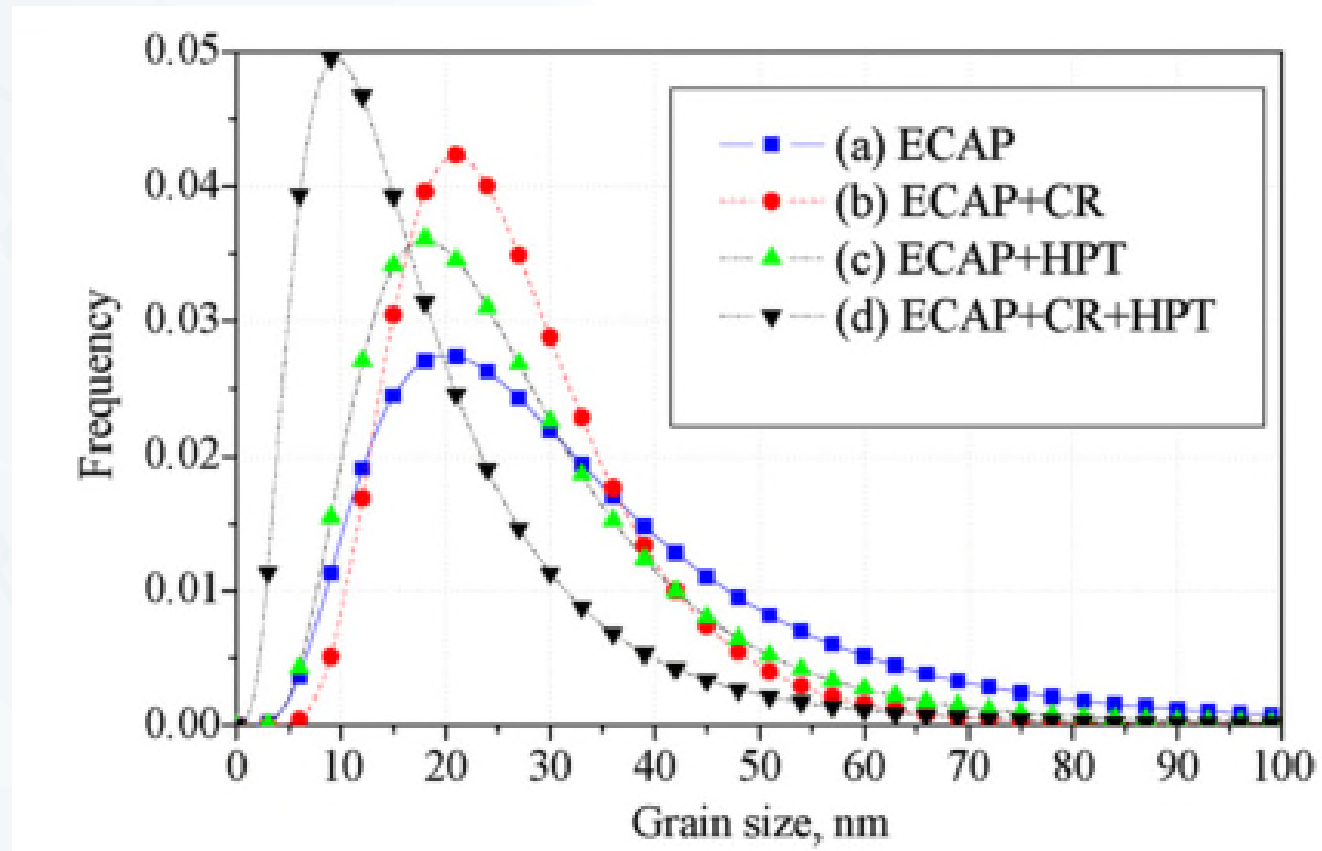
Grain size evolution

HPT opposite to ECAP

HPT same direction as ECAP



## Combined effect of ECAP + HPT



# Severe Plastic Deformation techniques potentials

# SPD potentials

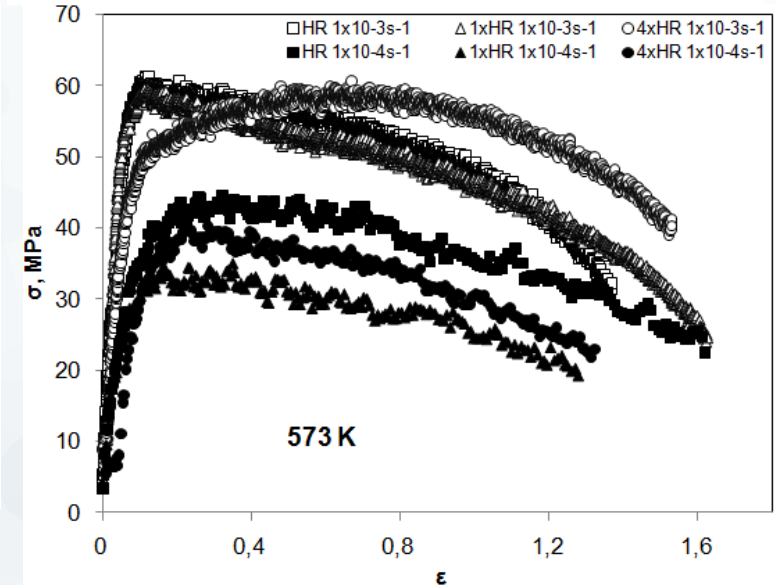
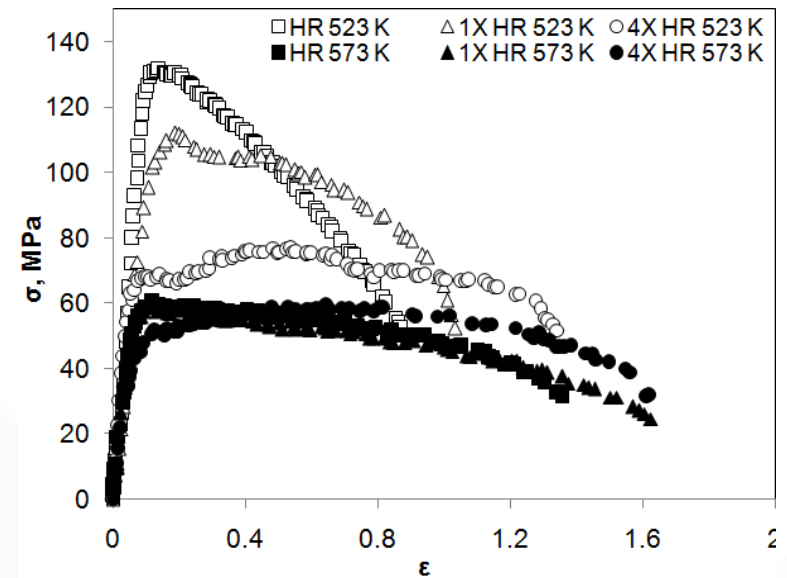
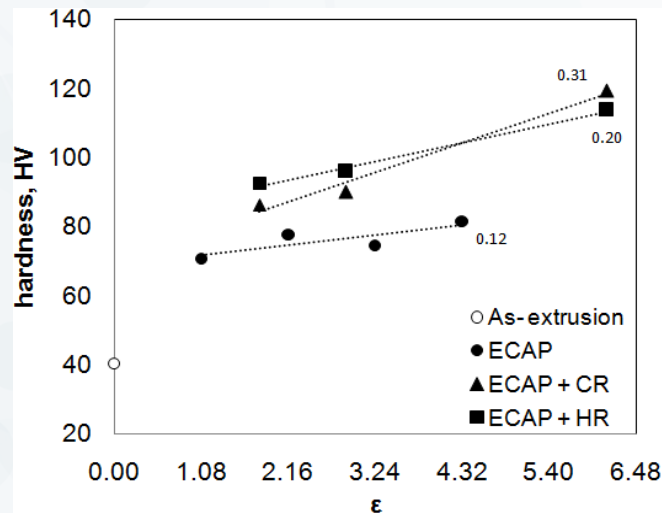


## ECAP for Ductility improvement

Tensile stress-strain curves at  $\dot{\epsilon}=1 \times 10^{-3} \text{ s}^{-1}$  of the AA3004 alloy (523 and 573K) (up)

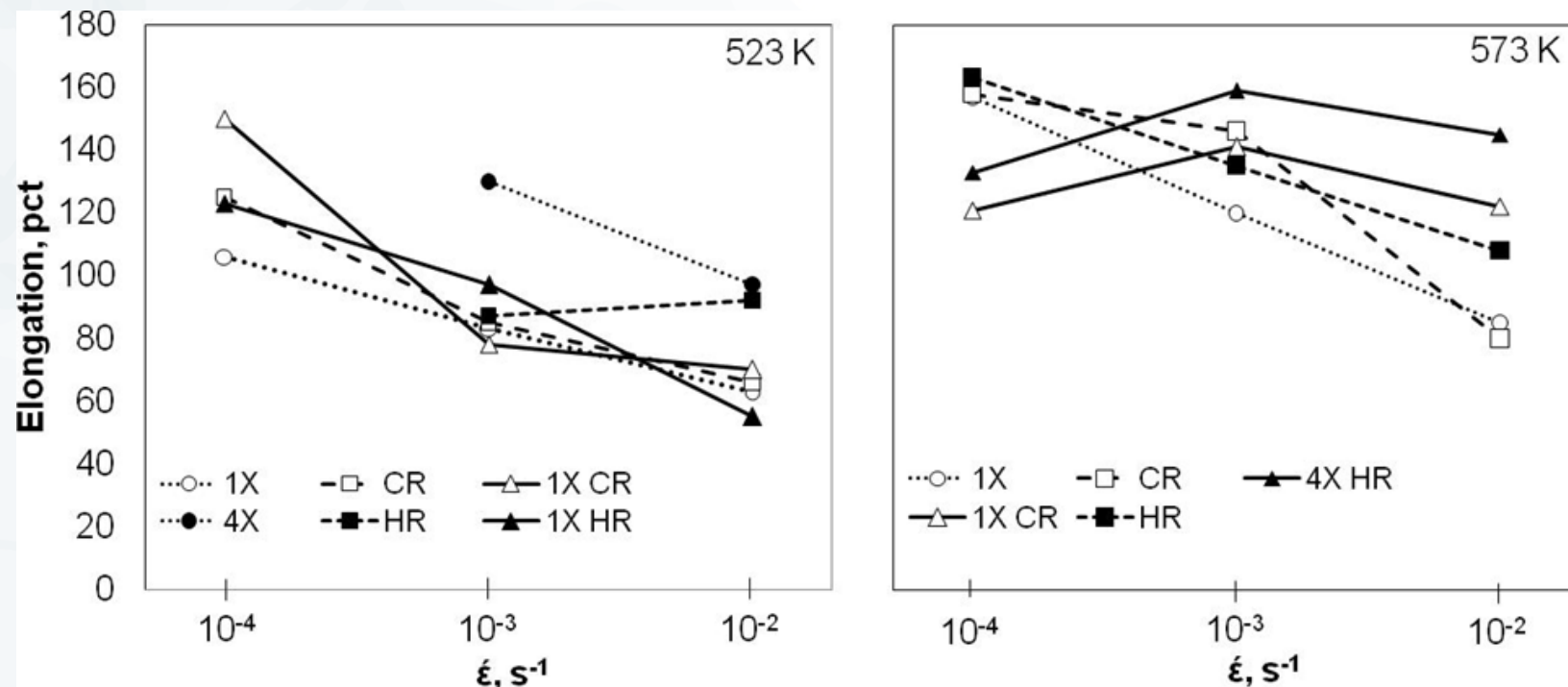
at 573K for decreasing strain rate ( $\dot{\epsilon}=1 \times 10^{-3} \text{ s}^{-1}$ ,  $\dot{\epsilon}=1 \times 10^{-4} \text{ s}^{-1}$ ) (523 and 573K) (down)

Hardness vs. accumulative strain for ECAP (1X-to-4X), ECAP+CR and ECAP+HR



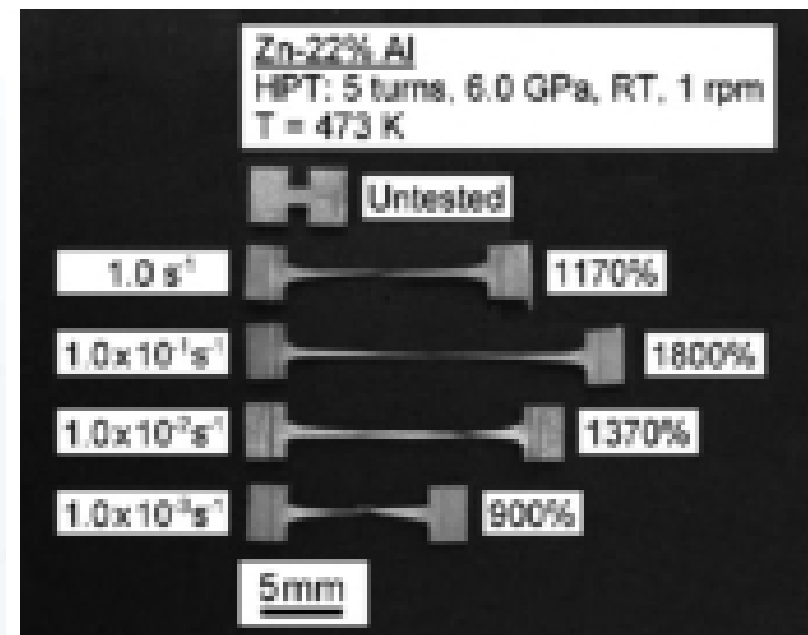
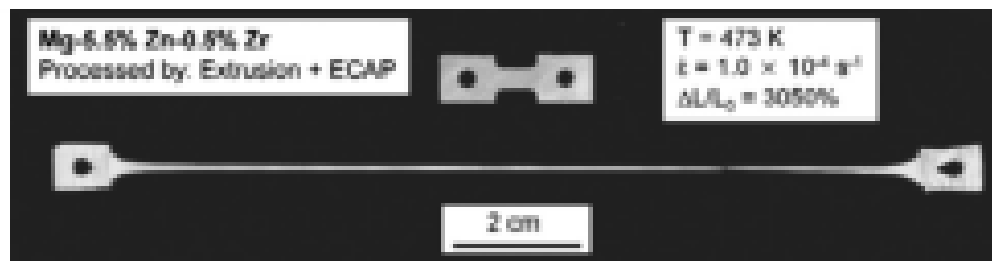
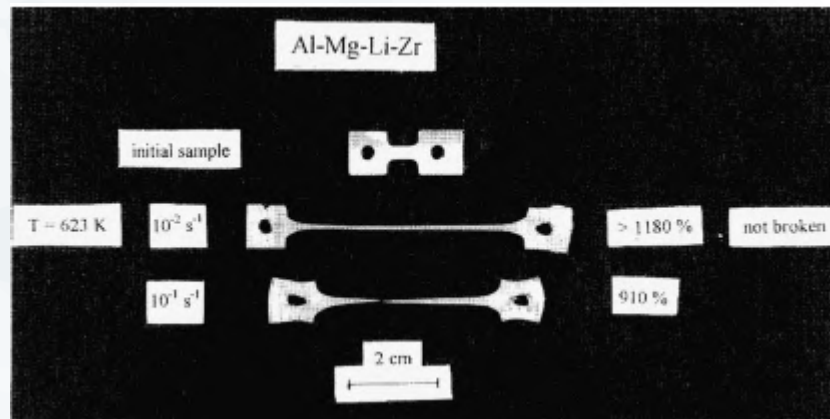
## ECAP for Ductility improvement

### A better sheet formability of AA3000 series / ECAP-route A



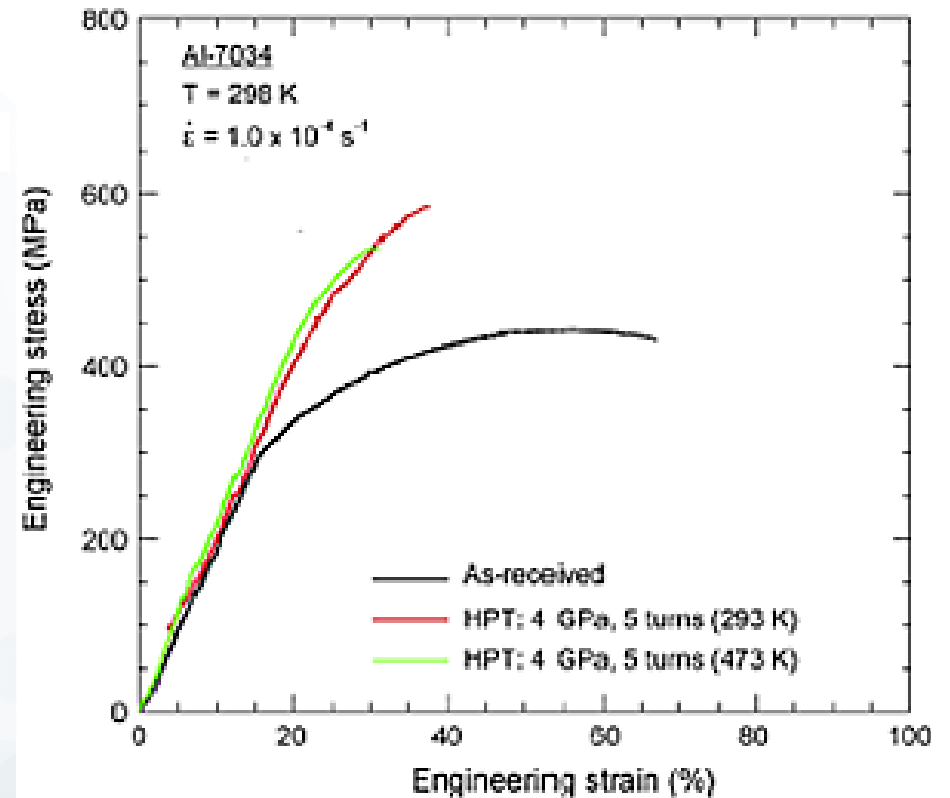
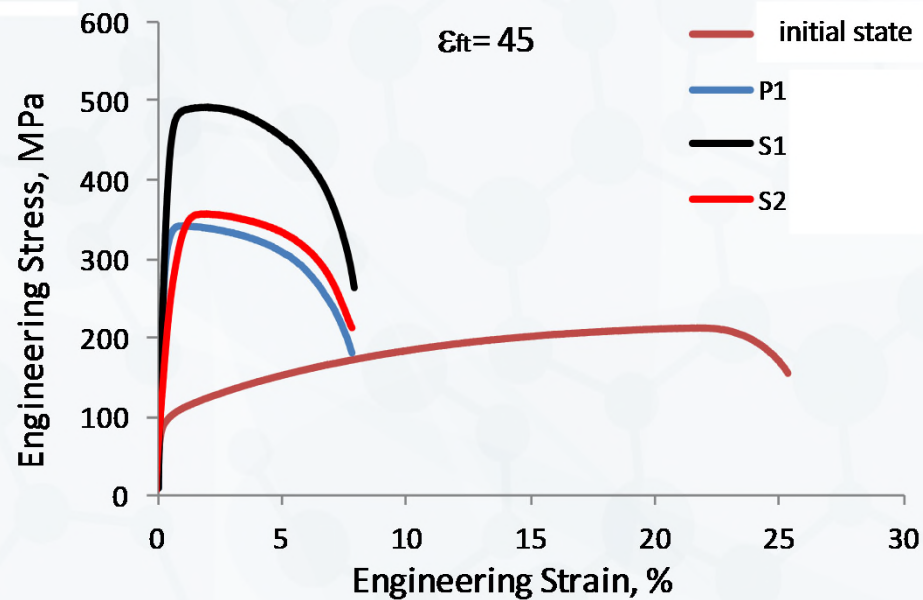
**best elongation to failure response: ECAP+HR and ECAP+CR at 573K**  
**with no significant ductility deterioration with strain rate**

## Superplasticity behavior



After T.C. Langdon  
T.G. LANGDON, Acta Mater., 61, (2013) p.7033.

## Superplasticity behavior

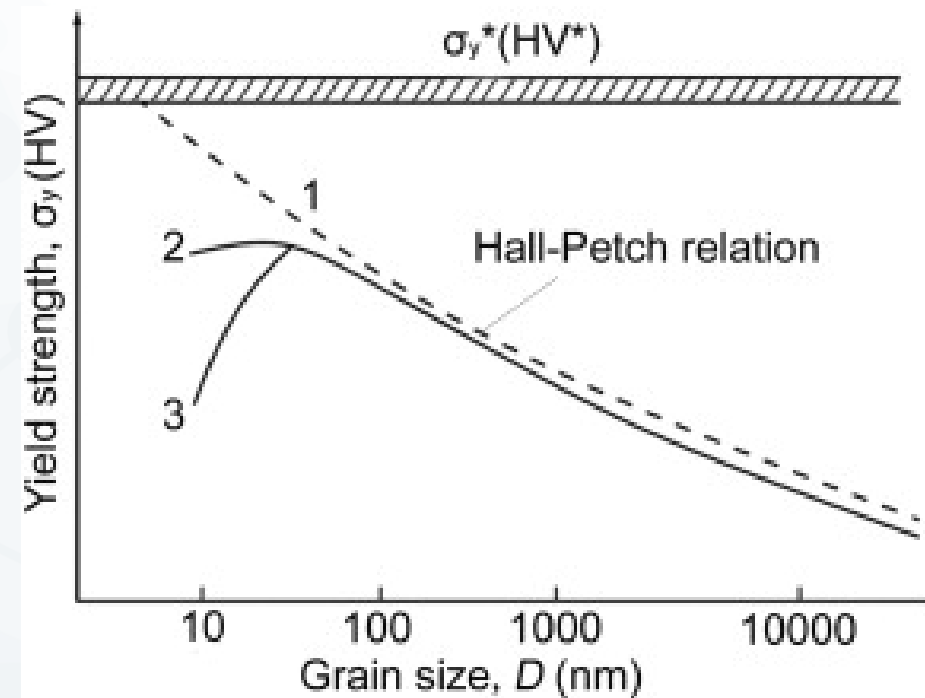
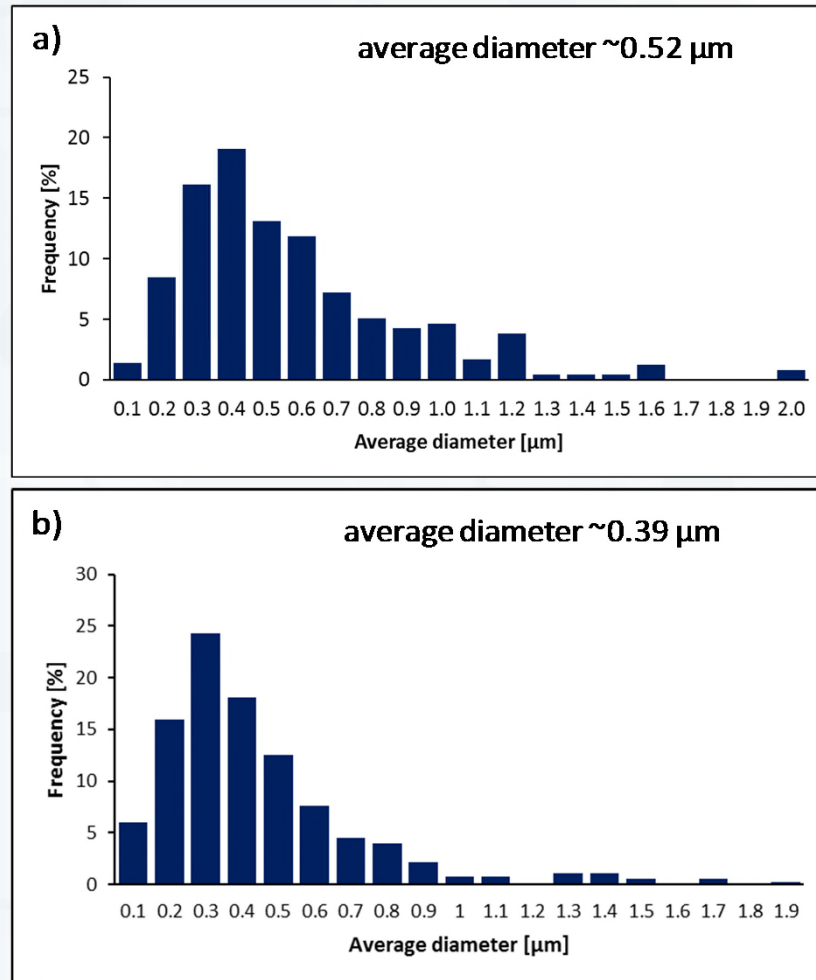


After T.C. Langdon  
T.G. LANGDON, Acta Mater., 61, (2013) p.7033.

# Limits of ECAP and HPT

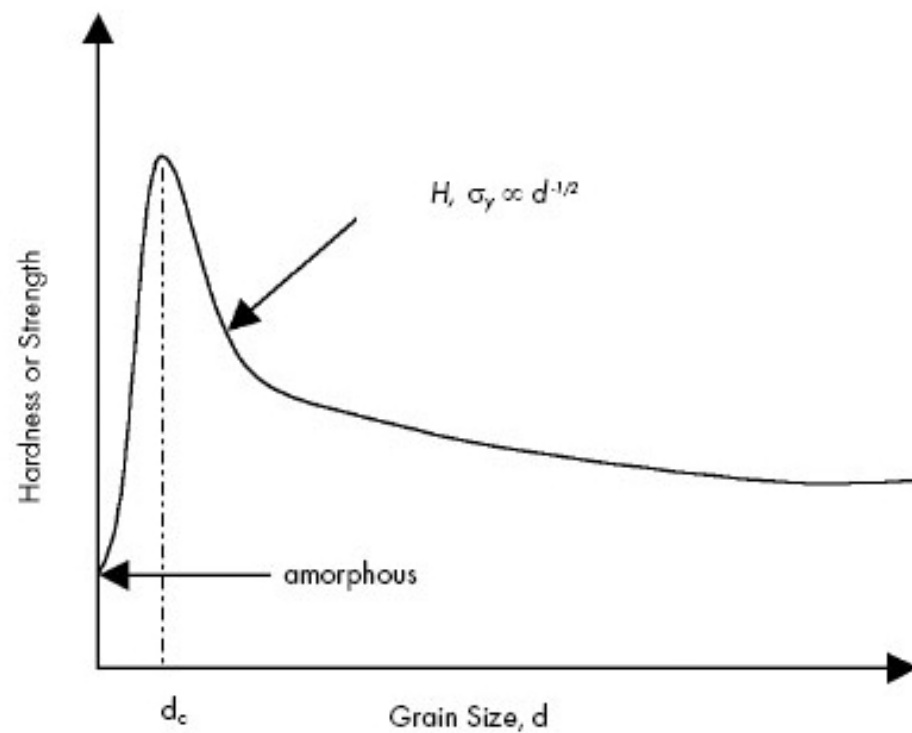
# Limits of ECAP and HPT

## grain size reduction physical limit



# Limits of ECAP and HPT

grain size reduction physical limit:  
deviation from the Hall-Petch relationship



## Hall-Petch Equations

$$\sigma_y = \sigma_o + Kd^{-1/2}$$

$$H = H_o + Kd^{-1/2}$$

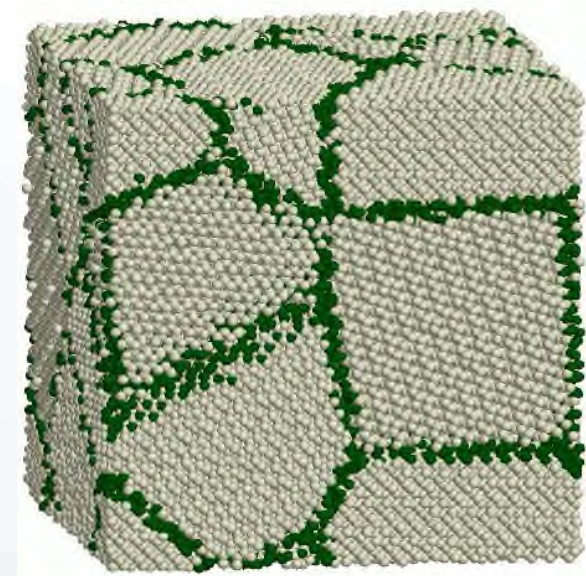
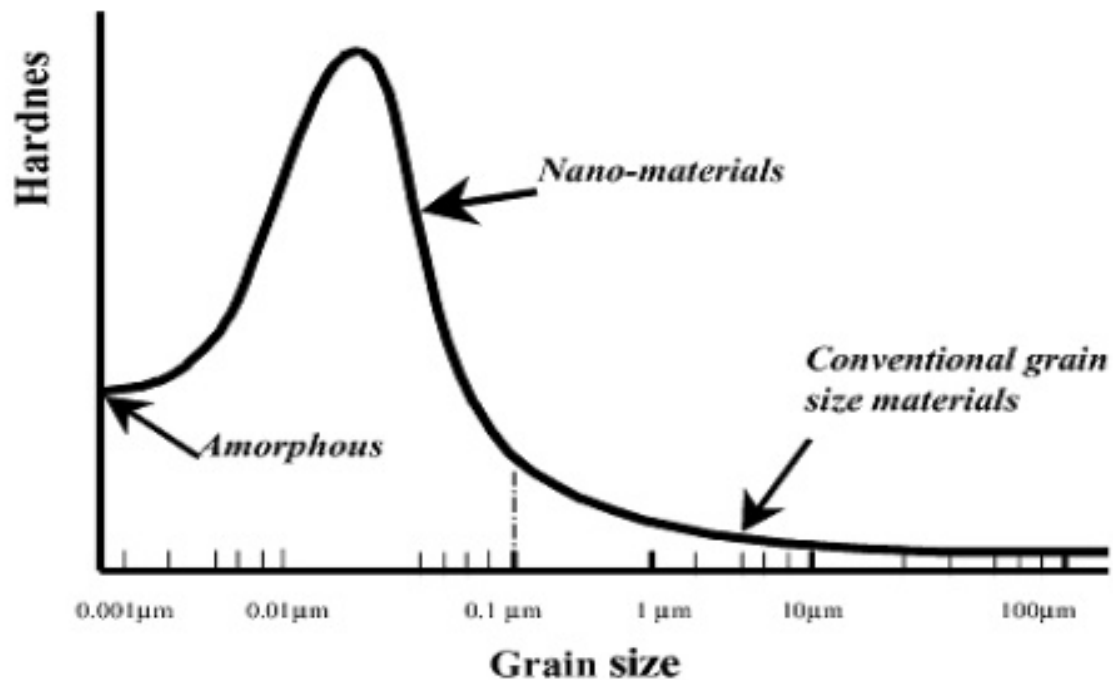
$\sigma_y$ : yield strength  
 $\sigma_o$ : lattice yield stress  
(minimum stress  
required to move  
a dislocation)

$H$ : hardness  
 $H_o$ : single crystal  
hardness

$K$ : constant  
 $d$ : average grain size

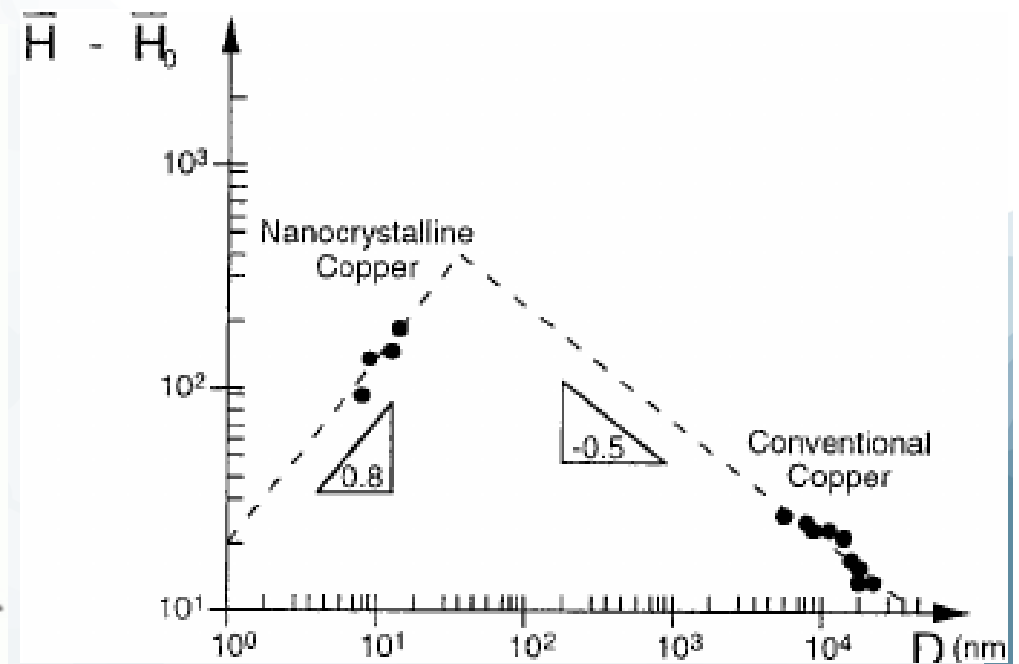
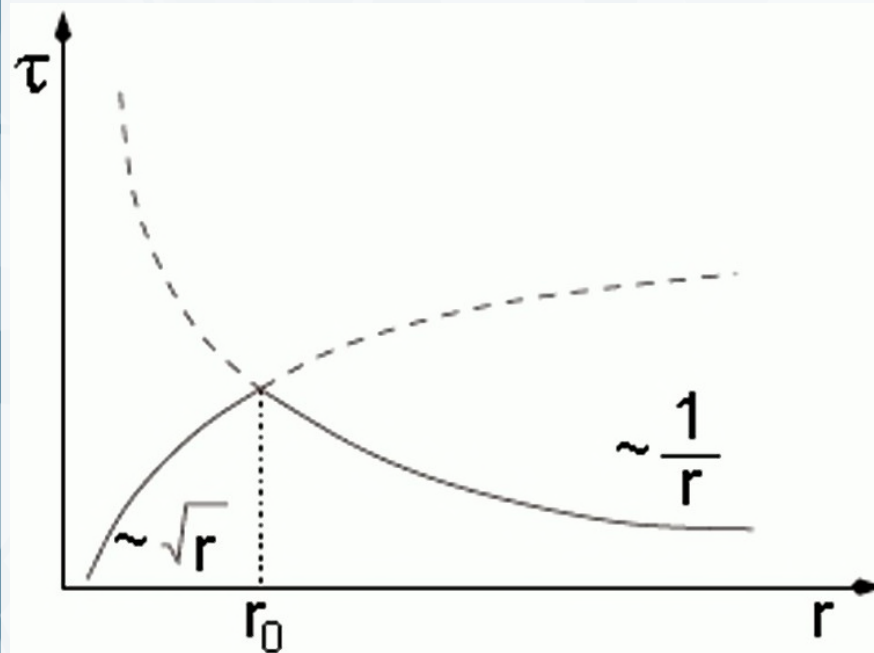
## Limits of ECAP and HPT

grain size reduction physical limit:  
deviation from the Hall-Petch relationship  
(grain boundary region role)

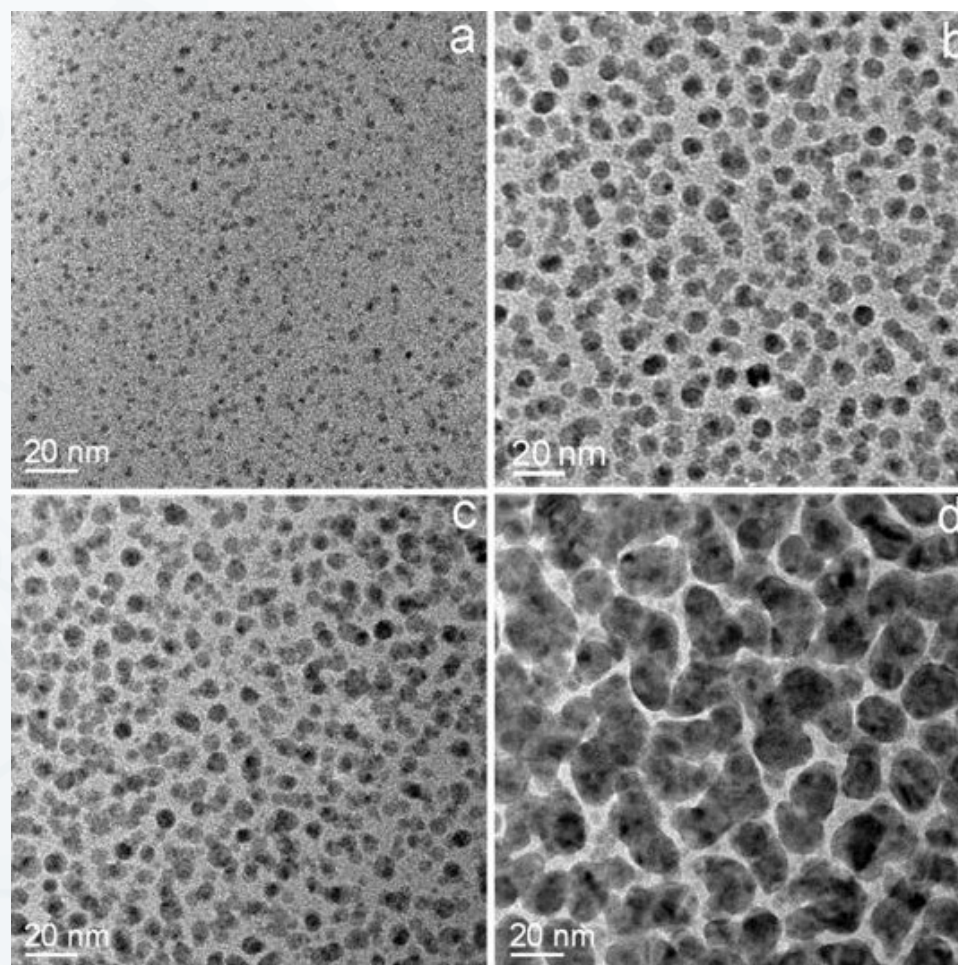


## Limits of ECAP and HPT

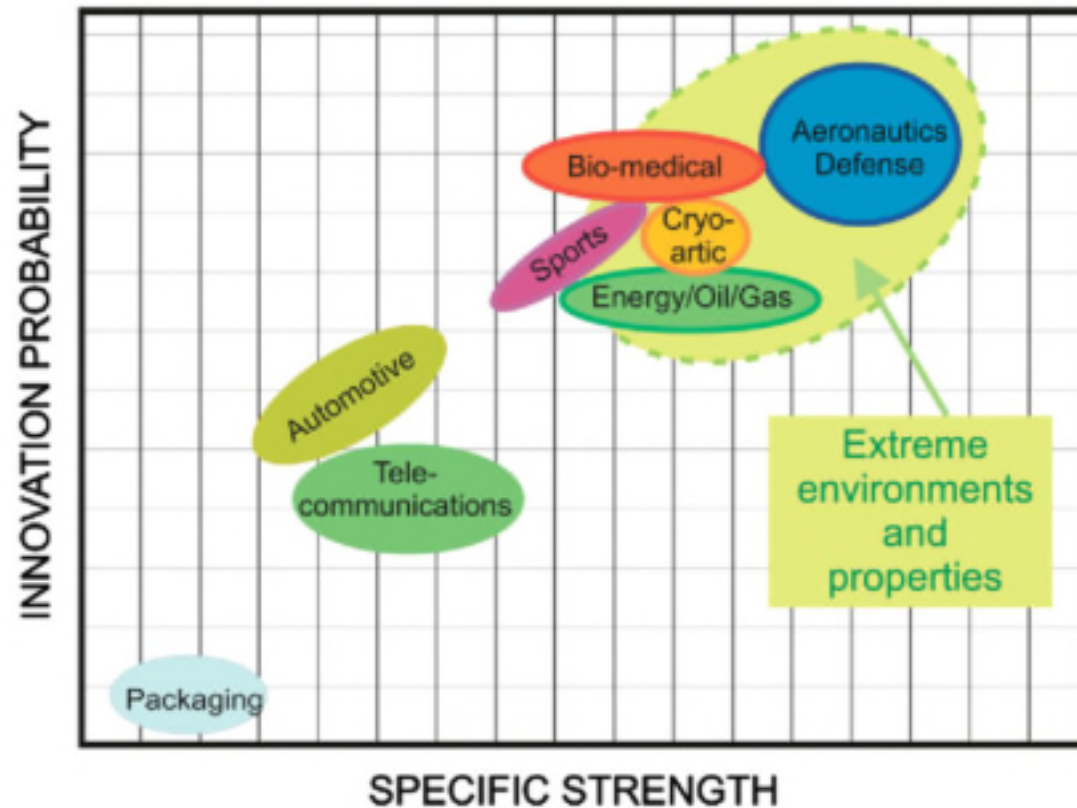
grain size reduction physical limit:  
deviation from the Hall-Petch relationship  
(grain boundary region role)



# From the «Top-Down» approach to the «Bottom-Up»



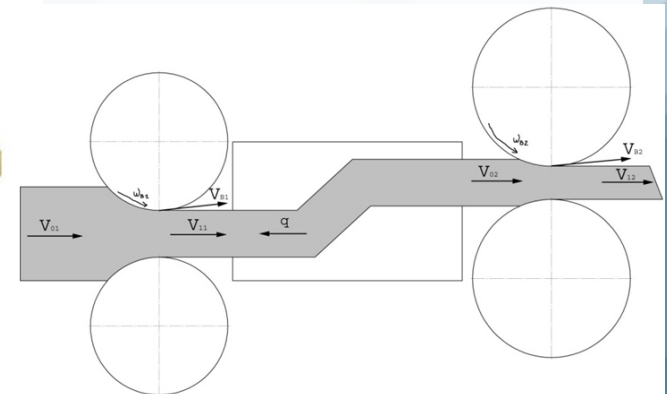
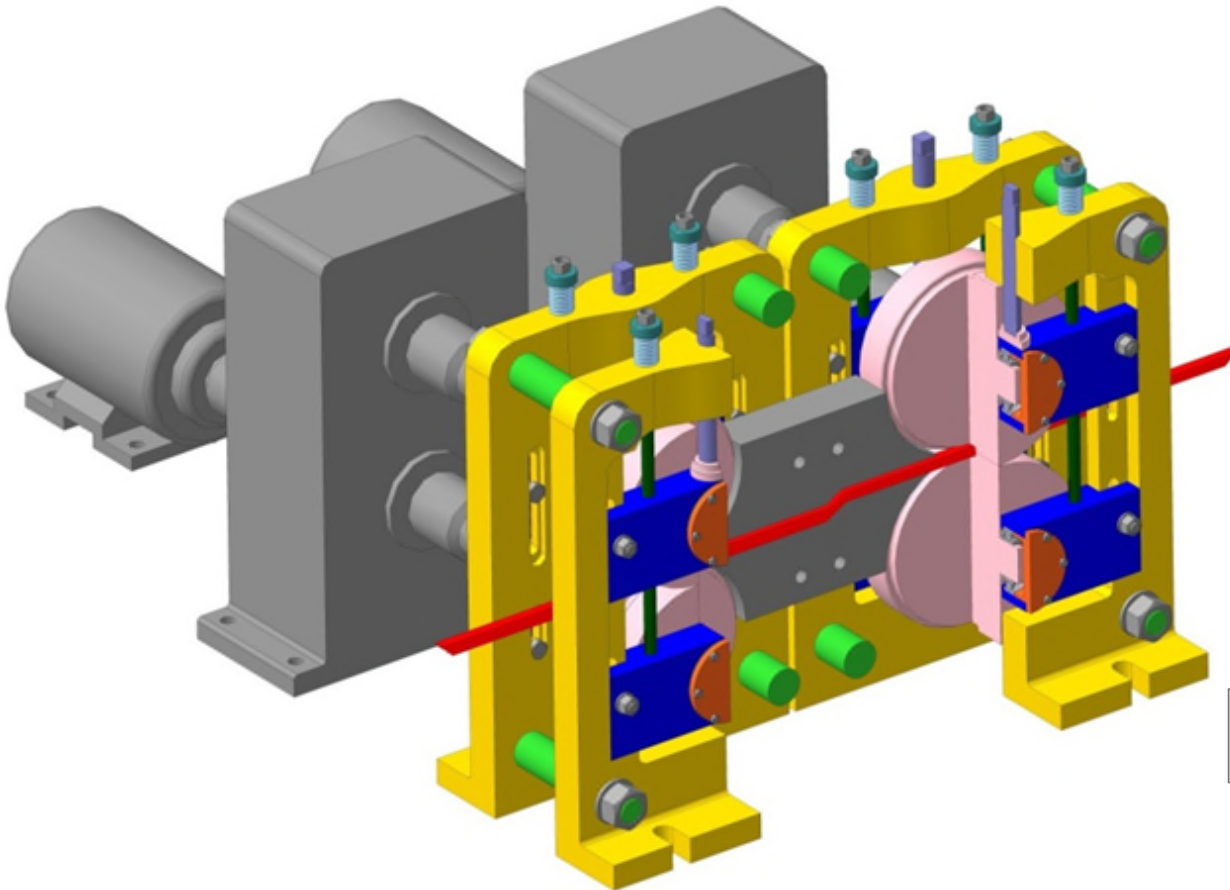
# OUTLOOK



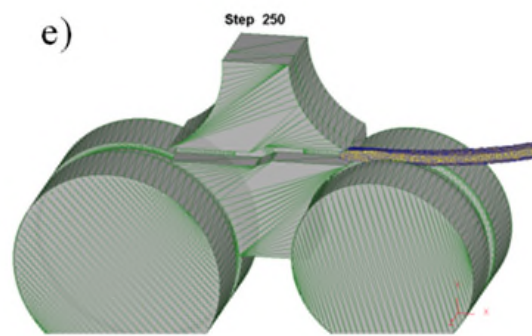
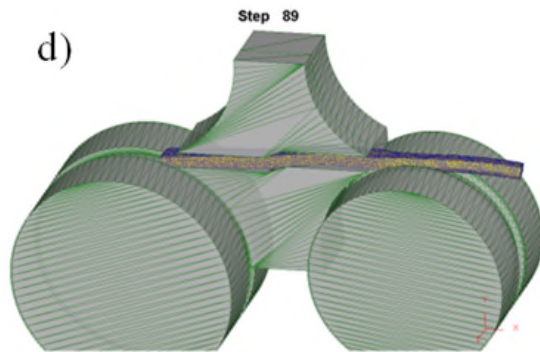
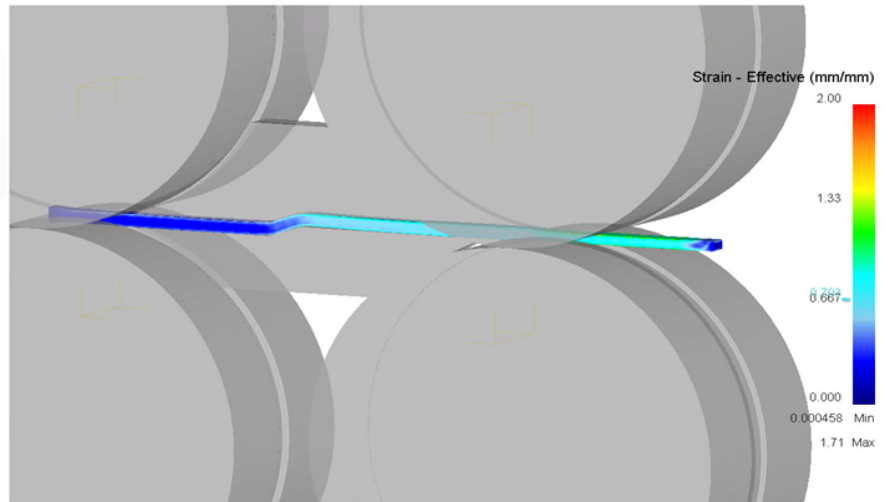
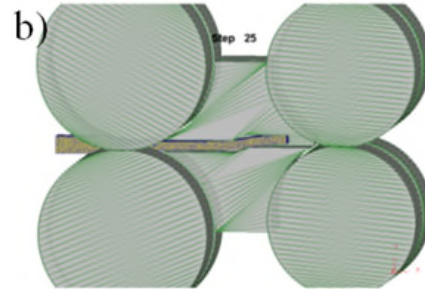
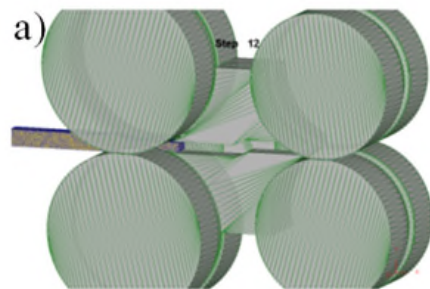
Innovation probability in various sectors versus the specific strength:  
the highest potential is anticipated in applications and products under extreme environments  
and/or requiring extreme specific strength

R.Z. Valiev , M.J. Zehetbauer, Y. Estrin, H.W. Höppel, Y. Ivanisenko, H. Hahn, et al. Adv Eng Mater, 9 (2007), 527

## Industrial scaling up



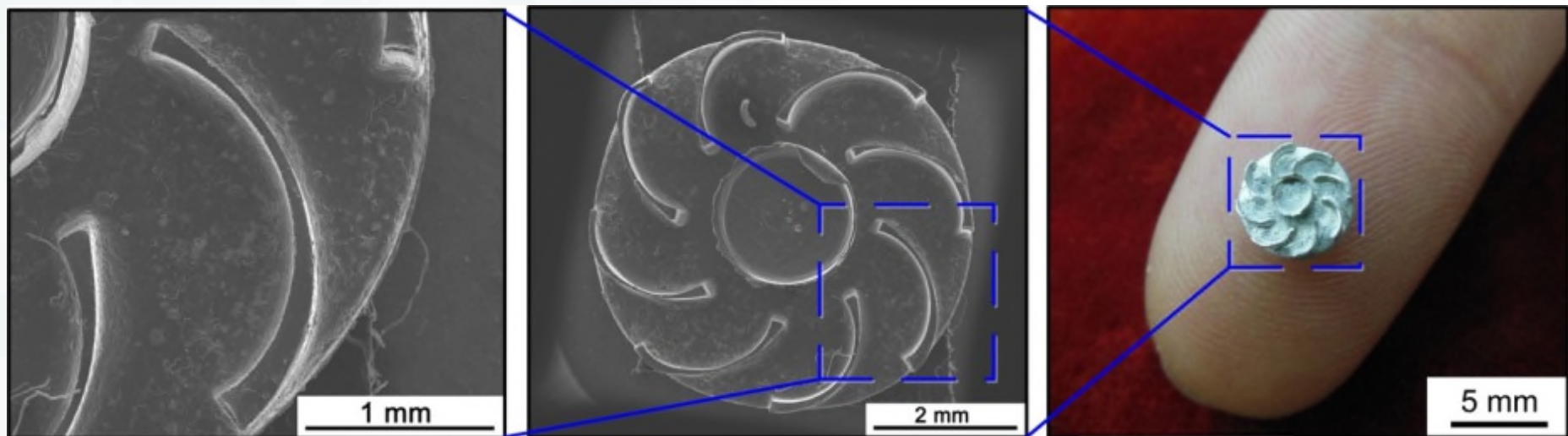
## Industrial scaling up



## Industrial scaling up: the prototype



## Micro-turbine of UFG pure aluminum formed at room- temperature



J. Xu, X. Zhu, L. Shi, D. Shan, B. Guo, T.G. Langdon. Adv. Mater. Eng., 17 (2015), 1022

## Marcello Cabibbo's Major Publications in the field of SPD on metallic materials

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**M. Cabibbo**, A TEM Kikuchi pattern study of ECAP AA1200 via routes A, C,  $B_C$ , Mater. Charact., 61, **2010**, 613.

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**M. Cabibbo**, *"Thermal stability study of ECA pressed AA1200 via route A, C, Bc"*, Rev. Adv. Mater. Sci. 25, 2, **2010**, 113.

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**M. Cabibbo**, *"Shear induced low- and high-angle boundary characterization using Kikuchi bands in transmission electron microscopy"*, Materials Science Forum Vols. 584-586: Nanomaterials by Severe Plastic Deformation IV, **2008**, 293.

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**M. Cabibbo**, E. Evangelista, M.E. Kassner, and M.A. Meyers, *"Transmission electron microscopy study of the strain induced low and high angle boundary development in equal-channel angular-pressed commercially pure aluminum"*, Ultrafine Grained Materials IV, Eds. Zhu, T.G. Langdon, Z. Horita, M.J. Zehetbauer, S.L. Semiatin, and T.C. Lowe. TMS (The Minerals, Metals & Materials Society), **2006**, 237.

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# Thank you!



The ICARUS project has received funding from the European Union's Horizon 2020 research and innovation programme under grant agreement No 713514.

The SUPERMAT project has received funding from the European Union's Horizon 2020 research and innovation programme under grant agreement No 692216.

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